

1 Thermal behavior study of pristine and modified halloysite nanotubes:
2 a modern kinetic study
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7 c aggiungere
8

9 **Abstract**

10 Pristine halloysite nanotubes (HNTs) were studied by thermogravimetry (TG) up to 800°C. Etching
11 of alumina from inside the tube (causing a significant increase of tube lumen) was realized by treating
12 the material with an acidic H₂SO₄ solution at 50°C. Both materials were characterized by TG-FTIR
13 techniques and their thermal behaviors were compared with that of kaolinite (KAO). The coupling
14 of TG with FTIR enables to detect the gases evolved during the TG experiments, thus confirming that
15 only pristine HNTs undergo dehydration with the loss of interlayer water molecules at around 245°C,
16 while dehydroxylation occurs in all these materials in close temperature ranges around 500°C. TG
17 runs at five different heating rates (2, 5, 10, 15 and 20 °C min⁻¹) were carried out in the same
18 experimental conditions used for the thermal analysis study with the aim to investigate dehydration
19 and dehydroxylation kinetics using some isoconversional methods, recommended by the ICTAC
20 kinetic committee, and a modulated thermogravimetry heating rate method. Finally, the results of the
21 kinetic analysis were discussed and explained in terms of the strengths of the hydrogen bonds broken
22 during these processes.
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24 **Introduction**

25 Halloysite is a two-layered aluminosilicate clay with chemical formula $\text{Al}_2\text{Si}_2\text{O}_5(\text{OH})_4 \times n\text{H}_2\text{O}$,
26 consisting of Si-tetrahedral outer sheets joined to Al-octahedral inner sheets by planes of oxygen
27 atoms of tubular shape. Its structure is made up of nano-size tubes having external diameter of 50-80
28 nm, cylindrical pore (lumen) of 10-15 nm and length of about 1000 nm [1]. Due to this characteristic
29 shape and the presence of less abundant surface hydroxyl groups (with respect to kaolin and
30 montmorillonite) halloysite nanotubes (HNTs) can be dispersed in polymers without the need of
31 exfoliation to form halloysite-polymer composites that are transparent in wide ranges of wavelengths,
32 including near-UV [1-4]. In addition, HNTs is not hazardous for the environment and is available in
33 large amount (thousand of tons) at a very low cost. The outer and inner structure is made up of polar
34 compounds (sheets of $[\text{SiO}_4]$ tetrahedra and sheets of edge-sharing $[\text{AlO}_6]$ octahedra), thus providing
35 a good hydrophilicity and therefore a good dispersion in polar polymers, like epoxy,
36 polyethyleneimine, polyamides, polyacrylates, polyvinyl alcohol and biopolymers, like starch, pectin,
37 chitosan, and humic acid [5-10]. Due to their elongated shape and cylindrical lumen, natural HNTs
38 can be loaded with several chemically and biologically active substances [4], like drugs [3,11],
39 proteins [3,12], DNA [13], antibacterials [14], cosmetics [15], thus providing useful
40 bionanocomposite materials for their controlled release for pharmaceutical applications. Sulfuric acid
41 treatments have been provided as an efficient method for enlargement of HNTs lumen diameter with
42 the aim to increase the tube loading capacity. The selective etching of alumina sheets was optimized
43 by tuning time, temperature, and acid concentration; in particular at high level of etching (above 30-
44 40% of dealumination) halloysite gradually loses its tubular morphology [16].

45 Nowadays it is not clear how HNTs are formed in nature. According to some authors, kaolinite
46 (KAO), with a layer structure consisting of superimposed silicon tetrahedral sheets and aluminum
47 octahedral sheets, is the main mineral phase of kaolin having the same chemical formula of halloysite.
48 It is demonstrated that KAO may roll leading to the formation of HNTs [17]. KAO is an important
49 material used in several industrial processes like food-processing industry, oil shale processing,

50 ceramic industry, as a pozzolanic material or as a filling agent, and for its use it is pre-heated at high
51 temperature (from 450 to 700°C) until it is transformed to metakaolin via dehydroxylation [18].
52 Dehydroxylation, elimination of water from hydroxyl groups, is an important thermal dissociation
53 reaction among those occurring in the kaolinite group minerals and in natural or synthesized silicate
54 materials [19]. Non-isothermal kinetics of this process, which occurs at temperature quite higher than
55 dehydration due to the presence of stronger hydrogen bonds between the OH groups, was extensively
56 studied for different type of KAOs [18,20-24], but little is known about the same process occurring
57 in HNTs where early papers used questionable methods under isothermal conditions [25,26].
58 Some authors adopted in their studies [21b-c,24] kinetic approaches (in particular, model-fitting
59 methods) that the ICTAC kinetics committee demonstrated to provide unreliable results [27].
60 Furthermore, Liu and co-workers recently applied an early method to study dehydroxylation of
61 synthetic Al-goethite, providing unreasonable low activation energies between 3 and 7 kJ mol⁻¹ [28].
62 Therefore, a first aim of this study is to investigate the thermal behavior of pristine and modified via
63 lumen enlargement HNTs and to compare it with that of KAO. A second aim is to provide an
64 exhaustive description of kinetic analysis of both dehydration (for HNTs only) and dehydroxylation
65 processes occurring in pristine and modified HNTs (treated at 50°C with sulfuric acid), with a view
66 to correlate the results obtained with the different structures of these materials. In particular the
67 coupling of thermogravimetry (TG) with FTIR enabled to completely characterize the thermal
68 degradation of HNTs (both pristine and etched) whereas TG experiments performed at different
69 heating rates were used for the kinetic study. The data so obtained were analysed by several
70 isoconversional methods and compared with those obtained by modulated thermogravimetry.

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73 **Theoretical background**

74 The kinetic description of thermally stimulated processes occurring in materials in the condensed
75 phase is rather more complex than that for homogeneous reactions. The first difficulty arises from the

76 definition of the function describing the progress of the reaction against time (under isothermal
 77 condition) or temperature (under the most commonly used non-isothermal condition with constant
 78 heating rate $\beta=dT/dt$). The so-called degree of conversion α , which is 0 at the initial temperature T_i
 79 and 1 at the final temperature T_f of each step, is defined as $\alpha=(m_i-m_T)/(m_i-m_f)$, where m_i , m_f and m_T
 80 are the sample masses at the corresponding T_i , T_f and T according to thermogravimetric (TG) data.
 81 The explicit dependence of the reaction rate by both the absolute temperature and the extent of
 82 conversion α , strictly valid under the assumption of a of a single-step process, is expressed by the
 83 following equation

$$84 \quad \frac{d\alpha}{dt} = k(T)f(\alpha) \quad (1)$$

85 where $k(T)$ usually represents the rate constant, commonly used in the form of the Arrhenius equation
 86 that enable to re-write Eq. (1) in the following form

$$87 \quad \frac{d\alpha}{dt} = Ae^{-E/RT}f(\alpha)$$

88 (2)

89 where A is the pre-exponential factor, E is the activation energy and $f(\alpha)$ is a function called reaction
 90 model [29,30].

91 For experiments carried out under modulated heating rate conditions using an oscillatory temperature
 92 program (temperature increases “smoothly varying temperature sine wave” [31]), the ratio between
 93 the reaction rates expressed in Eq. (2) and calculated at the peak and valley of the sinusoidal wave
 94 form (α_p and α_v), can be derived as follows:

$$95 \quad \frac{d\alpha_p/dt}{d\alpha_v/dt} = \frac{e^{-E/RT_p}f(\alpha_p)}{e^{-E/RT_v}f(\alpha_v)} \quad (3)$$

96 where the $\ln((\alpha_p/\alpha_v))$ signal is obtained using a Fourier transformation. Under constant conversion
 97 condition (or at least for small variation between adjacent peaks and valley) the ratio $f(\alpha_p)/f(\alpha_v)$
 98 approaches unity, being $f(\alpha_p)\approx f(\alpha_v)$. Taking the logarithm of both sides of Eq. (3) and solving for E
 99 yields:

100
$$E = \frac{RT_p T_v \ln(d\alpha_p/d\alpha_v)}{T_p - T_v} \quad (4)$$

101 In any oscillatory temperature program, T_p and T_v are defined as $T_p = \langle T \rangle + A$ and $T_v = \langle T \rangle - A$, where
 102 $\langle T \rangle$ is the average temperature, A is the temperature amplitude, while $T_p - T_v = 2A$. Eq. (4) can be
 103 further simplified by introducing a new parameter L , which is set equal to the peak-to-peak amplitude
 104 of the $\ln(d\alpha)$ signal ($L = \ln d\alpha_p - \ln d\alpha_v$). Once the values have been replaced in Eq. (4), it assumes
 105 the following more simple form:

106
$$E = \frac{R(T^2 - A^2)L}{2A} \quad (5)$$

107 It is worth noting that a kinetic method based on Eq. (5) is among those called “model-free”, as its
 108 computations do not depend on the knowledge of the choice of a proper model function $f(\alpha)$.

109 On the other hand, approaches based on multiple heating rate (or temperature) programs are highly
 110 recommended by the ICTAC Kinetic Committee [32]. The time dependency of reaction rate $d\alpha/dt$ is
 111 then replaced by its corresponding temperature dependency $d\alpha/dT = \beta^{-1} d\alpha/dt$ reaction rate, giving
 112 Eq. (6):

113
$$\frac{d\alpha}{dT} = \left(\frac{A}{\beta}\right) e^{-E/RT} f(\alpha)$$

 114 (6)

115 Separation of variables in Eq. (6) yields:

116
$$\frac{d\alpha}{f(\alpha)} = \left(\frac{A}{\beta}\right) e^{-E/RT} dT$$

 117 (7)

118 The integrals of both the left- and right-hand side of Eq. (7) gives:

119
$$\int_0^\alpha \frac{d\alpha}{f(\alpha)} = g(\alpha) = \left(\frac{A}{\beta}\right) \int_0^T e^{-E/RT} dT$$

 120 (8).

121 The temperature integral in Eq. (8) has no exact analytical but approximate solutions that give rise to
 122 some of the most commonly isoconversional methods, whose equations for each fixed extent of
 123 conversion α have the following general form:

124
$$\ln\left(\frac{\beta}{T^B}\right)_\alpha = Const - C\left(\frac{E}{R}\right)\left(\frac{1}{T\beta}\right)_\alpha \quad (9)$$

125 where B and C are adjustable parameters, whose values depend on the approximation made. In
 126 particular, for the Ozawa-Flynn-Wall (OFW) method [33,34], based on the Doyle's approximation
 127 [35], $B=0$ and $C=1.052$. More accurate results can be obtained using the Kissinger-Akahira-Sunose
 128 (KAS) method [36], where $B=2$ and $C=1$ or the Starink (STA) method [37] ($B=1.92$ and $C=1.008$).
 129 Each value of activation energy at each given extent of conversion is calculated from the slope of the
 130 regression line obtained by plotting the left-hand side of Eq. (9) against the reciprocal temperature
 131 $(T\beta^{-1})_\alpha$. Isoconversional methods, along with the invariant kinetic parameters [38,39] and the constant
 132 rate thermal analysis (CRTA) [40,41], are recognized to be among those who usually give reliable
 133 results and relevant books, review and papers [27,29,42,43] deal with the advantage of their use.
 134 These methods are based on the assumption that the reaction rate at constant degree of conversion is
 135 only a function of temperature. Vyazovkin developed a method (VYA) that gives results with a better
 136 accuracy by numerical integration of the right-hand side of Eq. (5) [44-46] by minimizing the
 137 following function:

138
$$\phi(E_\alpha) = \sum_{i=1}^n \sum_{j \neq i}^n \frac{J[E_\alpha, T_i(t_\alpha)]}{J[E_\alpha, T_j(t_\alpha)]} \quad (10)$$

139 where $J[E_\alpha, T(t_\alpha)] = \int_0^{T_\alpha} \exp[-E_\alpha/RT(t)]dT$ is solved numerically and minimization is made for each
 140 value of α with the aim to obtain a conversion dependency of activation energy. The reaction model
 141 and the α -dependence of pre-exponential factor ($\ln A_\alpha$) can be accurately determined only in the case
 142 of processes that follow approximately a single-step kinetics, for which it can be expected that
 143 activation energy does not varies appreciably over the entire range of the extent of conversion α by
 144 combining the results of isoconversional (model-free) and model-fitting methods [47]. By applying
 145 a model-fitting method (Coats-Redfern [48] in this study) a pair of Arrhenius parameters can be
 146 obtained for each reaction model using a single-heating rate experiment. Wide ranges of values are
 147 found for both parameters when all the reaction models are considered, but a strong linear correlation

148 denoted as compensation effect is found between them in the following form:

$$149 \quad \ln A_i = aE_i + b \quad (11)$$

150 where the subscript i refers to each of all the reaction model. Once a and b parameters have been
151 determined at each heating rate using a linear regression procedure, these values were replaced in Eq.
152 (11) by their mean values $\langle a \rangle$ and $\langle b \rangle$ while the E_i values were replaced by the isoconversional
153 values of E_α to determine the corresponding values of $\ln A_\alpha$ for each given value of α [47].

154

155

156 **Experimental**

157 **Materials**

158 Kaolinite and pristine halloysite nanotubes were purchased from Sigma Aldrich and used without
159 further purification. The procedure followed for the HNT lumen etching was similar to that reported
160 in literature [16]. A suspension of halloysite was obtained by dispersing 5 g of Halloysite in 300 ml
161 of a 2 mol l⁻¹ H₂SO₄ solution. The suspension was magnetically stirred for 48 hours on a hot plate at
162 the controlled temperature of 50 °C. The processed halloysite was then washed with distilled water
163 until the pH of the supernatant from the washing stage was in the range 6-7, similar to that of pure
164 halloysite suspension. The sample was dried in an oven at 50°C and then characterized by TG.

165

166 **Instruments**

167 Samples were analyzed through scanning electron microscopy (SEM) and scanning transmission
168 electron microscopy (STEM). Images were collected using an Ultra High Resolution Field Emission
169 Scanning Electron Microscopy (UHR-FE-SEM) by Zeiss equipped with a STEM. In order to collect
170 SEM images, powder of pristine and etched HNTs have been deposited onto a substrate of copper.
171 Instead the STEM samples were prepared using as substrate a TEM grid.

172 All the TG experiments were performed using a TA Instruments Q5000IR thermogravimetric
173 instrument equipped with an FTIR (Agilent Technologies) Cary 640 spectrophotometer for Evolved

174 Gas Analysis (EGA). The Q5000IR thermogravimetric analyzer has the optional capability to work
175 in modulated mode (modulated thermogravimetry, MTG). TG measurements were performed in this
176 study with both dynamic conventional (constant heating rate) and modulated temperature programs.
177 Both types of experiments were carried out from 40 to 800 °C using Pt crucibles under a stream of
178 air of 25 ml min⁻¹. In conventional TG, samples were heated at five different heating rates (2, 5, 10,
179 15, 20 °C min⁻¹) to process TG data for kinetic computations. In MTGA experiments samples were
180 heated at a heating rate of 2° C min⁻¹ with a temperature modulation amplitude of ± 5°C and a period
181 of 200 s. The MTG curves were analyzed using the TA Universal Analysis 2000 software. TG-FTIR
182 measurements were performed at a rate of 20 °C min⁻¹, from 40 °C to 800 °C under air flow (70 ml
183 min⁻¹), from 600 to 3000 cm⁻¹ with a 4 cm⁻¹ width slit. To reduce the background absorption from
184 water and carbon dioxide in the atmosphere, the optical bench was purged with nitrogen. In addition,
185 a background spectrum was taken before each analysis in order to zero the signal in the gas cell and
186 to eliminate the contribution due to the amount of ambient water and carbon dioxide. The amount of
187 sample in each TG and TG-FTIR measurement varied between 4 and 8 mg.

188

189 **Results and discussion**

190 Treatments with sulfuric acid have been provided as an efficient method for enlargement of HNTs
191 lumen diameter with the aim to increase the tube loading capacity [16]. The increase of the HNTs
192 lumen, after acidic treatment, is clearly evident in Fig. 1, where it is possible to see some examples
193 of SEM and STEM images of pristine and etched HNTs. STEM combines the principles of
194 transmission electron microscopy and scanning electron microscopy. Its primary advantage over
195 conventional SEM imaging is the improvement in spatial resolution with consequently better imaging
196 resolution.

197 Fig. 1a-b shows the STEM images of pristine and etched HNTs, respectively. The arrows and the
198 dashed lines highlight the size of the HNT lumen, which, in the case of pristine HNTs, is 15-20 nm;
199 while in etched HNTs, because of the etching process, increases up to 30-40 nm. The enlargement of

200 lumen size is further supported by SEM images (Fig. 1c-d). As could be expected the images of
201 etched HNTs (Fig. 1b,d) show changes in halloysite morphology. Indeed, although the rodlike
202 structure was preserved, the etched tubes present broken points and the halloysite walls appear more
203 friable and porous.

204 The TG/DTG curves of pristine HNTs, etched HNTs and KAO are given in Fig. 2. Relevant data
205 taken from these measurements (peak DTG temperatures and mass loss percentages) of each step are
206 shown in Table 1. The thermal behavior of HNTs and etched HNTs showed remarkable differences.
207 In particular, pristine HNTs undergo four steps of mass loss, the first of which at $T_p=37.5$ °C (mass
208 loss 2.3%) is due to water physically adsorbed to the surface, while the second, in the range 200-285
209 °C (mass loss 3.3%), is ascribed to the release of interlayer water molecules bound by hydrogen
210 bonds. The third step is due to dehydroxylation (condensation of hydroxyl groups of aluminum inner
211 sheets) around 470°C (mass loss 11.4%), while at 744°C the mass loss of 1.7% is ascribed to the
212 release of SO₂ due to the thermal decomposition of sulfides (as impurity) or alunite, according to
213 what it was recently reported in literature [20,49], even for kaolin [50]. The hypothesis that the mass
214 losses below 500°C could be exclusively due to water release, is confirmed by FTIR analysis. The
215 gas evolved at the fixed temperatures of 37.5, 245 and 468 °C present the same spectrum (see Fig.
216 3a) showing the narrow bands at 4000-3500 and 2000-1300 cm⁻¹ typical of the spectrum of water
217 [51]. On subsequent heating, the spectrum recorded at 750°C (Fig. 3b) shows the characteristic bands
218 of SO₂ impurities (1390, 1338 and 1180 cm⁻¹) [51, 52].

219 By contrast, etched HNTs did not show the loss of interlayer water around 245°C, thus demonstrating
220 that the acidic etching at 50°C, responsible for the enlargement of HNTs lumen [16], caused the
221 elimination of the interlayer water. The water loss attributed to dehydroxylation takes place at
222 temperature slightly higher (T_p at 479°C instead of 468 °C), and with a lower mass loss (6.0% instead
223 of 11.4%) than in pristine HNTs. As expected, around 730°C the loss of SO₂ is confirmed also in
224 etched HNTs. No dehydration is shown for KAO that is stable up to 330 °C in agreement with the
225 results reported in [53], while other authors found in the temperature range up to 150°C a slight mass

226 loss due to dehydration [20,21c]. Dehydroxylation occurs in a single step in the wide temperature
227 range between 400 and 700°C in agreement with literature findings [18,20,21c,22,50].

228 Eqs. 10 and 11 were considered to apply the four isoconversional methods denoted as OFW, KAS,
229 STA and VYA to analyze the kinetics of dehydration in HNTs by processing TG data between 200
230 and 280°C (Fig. 2), and dehydroxylation in HNTs, etched HNTs and KAO in the temperature range
231 close to 470-500°C. The results of kinetic analysis regarding dehydration and dehydroxylation,
232 reported as the usual conversion dependency of activation energy, are summarized in Figs. 4 and 5,
233 respectively. Interpretation of these results should be made in terms of the energy barrier to be
234 overcome by water molecules to proceed with dehydration or dehydroxylation. Application of OFW,
235 KAS and STA methods to dehydration of HNTs seems to fail due to the significant change in E_{α}
236 values preventing the use of Eq. (7) that implies separation of variables, and is rigorously valid only
237 if neither the model function $f(\alpha)$ depends on temperature nor activation energy E_{α} on the degree of
238 conversion. Furthermore, these results indicate a multi-step nature of the process investigated.
239 However, the results of the VYA method, which can be applied even in the case of remarkable
240 variation of E_{α} values, are in close agreement with those determined with the previous cited methods
241 (in particular in the range $0.55 < \alpha < 0.95$, where E_{α} drops from 170 to 38 kJ mol⁻¹). It is worth noting
242 that these values are markedly higher than the molar vaporization enthalpy of water (≈ 44 kJ mol⁻¹).
243 On the other hand, the temperature range for the occurrence of this process (200-280°C) is remarkably
244 higher than that of pure water. At this regard, water vaporization kinetics from bulk and from clays
245 was recently investigated [54] and average values ranging from 43.8 to 56.2 kJ mol⁻¹ were obtained
246 for the former and the latter materials, respectively, because of the low temperature range of the
247 occurrence of this process (from -20 to 90°C).

248 However, at the beginning of dehydration process of HNTs ($\alpha < 0.30$) the E_{α} values obtained by the
249 VYA method increase from 180 to 200 kJ mol⁻¹ and then decreases to about 175 kJ mol⁻¹. Actually,
250 these change of E_{α} values are not so high (slightly higher than the associated uncertainties) even if

251 they cannot be neglected. In general, increasing and decreasing trends of E_α values should be
252 interpreted in terms of the increase or decrease of the energy barrier for the occurrence of the process
253 examined. A decrease followed by an increase of water molecular mobility occurring during heating
254 can be probably the main factor that generated those variations of E_α values. On the other hand, water
255 molecular mobility is particularly affected by the rupture of weaker hydrogen bonds within the $[AlO_6]$
256 octahedra inner sheets. Constant E_α values ($\approx 171 \text{ kJ mol}^{-1}$) were found in the range $0.30 \leq \alpha < 0.60$,
257 followed by a significant decrease to the values usually considered for free vaporization of pure water
258 ($40\text{-}45 \text{ kJ mol}^{-1}$). The decreasing trend of E_α values at the end of the process can be explained by the
259 fact that the water molecular mobility increases up to values comparable to those found in bulk pure
260 water. The results obtained with MTG, which adopted a differential isoconversional approach with
261 respect to the integral one offered by OFW, KAS and STA, is completely in disagreement with those
262 of all other methods considered at the beginning ($\alpha < 0.30$) and at the completion ($\alpha > 0.80$) of the
263 process. A reasonable explanation of these large deviations can be ascribed to the approximation
264 made in Eqs. (3-4) to consider $f(\alpha_p) \approx f(\alpha_v)$, which is reasonably valid in most of the conversion ranges.
265 However, this approximation is inapplicable for n-th order models when the values of α approach
266 unity and for the autocatalytic model function when α approaches either zero or unity (as in the case
267 of dehydration of HNTs examined in this study) [31]. Slightly higher values of E (of about 40 kJ mol^{-1})
268 are also found in the range $0.30 \leq \alpha < 0.80$, which are practically constant, similarly to what it is
269 observed using all the other methods.

270 In Fig. 5a substantial agreement is found among the isoconversional dependencies of E_α values
271 related to dehydroxylation determined using the five methods and the slight differences between the
272 results of the first three integral approaches (OFW, KAS and STA) and those of VYA and MTG are
273 limited to small ranges of α values. Large deviations of E values calculated using the MTG method
274 when α approaches zero and unity is attributed (as in the case of dehydration of HNTs) to the failure
275 of the approximation $f(\alpha_p) \approx f(\alpha_v)$ when a process (as probably in this case) can be described by a

276 autocatalytic model function [31]. Dehydroxylation in HNTs showed a trend similar to that of
277 dehydration in the range up to $\alpha=0.40$, followed by practical constant E_{α} values (around 190-200 kJ
278 mol⁻¹), in close agreement (within a usual estimated uncertainty around 5-7%) with the single value
279 found in literature (≈ 185 kJ mol⁻¹) [26]. As expected, quite higher E_{α} values are shown in the case
280 of etched HNTs (Fig. 5b) and good agreements are found among the E_{α} values determined with the
281 five methods applied (except in the case of the MTG method for $\alpha \approx 0$ and $\alpha \approx 1$). An increasing trend
282 of E_{α} values can be attributed to a decreasing trend in molecular mobility of water (obtained by
283 condensation of hydroxyl groups) during the course of the process, probably caused by the etching
284 of [AlO₆] inner sheets. At a first sight, dehydroxylation of KAO (Fig. 5c) seems to be described in a
285 more simple way, with superimposable and practically constant values of E_{α} determined by the five
286 methods (around 220 kJ mol⁻¹) for $\alpha < 0.6$. A good agreement (at least in the range $\alpha < 0.70$) is found
287 also with literature data on KAO [21c] and with the values determined for etched HNTs. Remarkable
288 deviation of the E_{α} values determined with the MTG method is shown in the range $\alpha > 0.8$, ascribed
289 to the above-cited failure of the approximation $f(\alpha_p) \approx f(\alpha_v)$, thus suggesting in this case that the
290 mechanism of dehydroxylation of KAO can be reasonably described using n-th order models [31].
291 These findings, along with the comparison of their thermal behavior, suggest that the lumen
292 enlargement caused by the acidic etching of the inner sheets of aluminum oxide provides a material
293 even more similar to KAO than pristine HNTs.

294

295

296 **Conclusions**

297 Coupling of TG and FTIR techniques demonstrated in this study to be a useful tool to investigate the
298 thermal behavior of both pristine and etched HNTs. The acidic etching of the [AlO₆] octahedra inner
299 sheets of pristine HNTs is a common procedure used to enlarge the cylindrical lumens in order to
300 load HNTs with suitable amounts of chemically and biologically active substances. SEM and STEM

301 techniques used in this study confirmed to be able in providing images proving indubitably that the
302 etching of HNTs produced changes in halloysite morphology. Although the rodlike structure was
303 preserved, the etched tubes presented broken points and the halloysite walls appeared more friable
304 and porous. These structural changes are responsible of relevant changes in the thermal behavior of
305 HNTs (*i.e.*, lost of layered water molecules) that appears more similar to kaolinite (tested only for
306 comparison purpose) than to precursor. A modern kinetic analysis of both dehydration and
307 dehydroxylation processes was performed in accordance with recent ICTAC recommendations.
308 Different differential (MTG) and integral (OFW, KAS, STA and VYA) isoconversional methods
309 confirmed the complex multi-step nature of both processes, evidenced by the non-negligible variation
310 of E_{α} with increasing the degree of conversion. Increasing and decreasing trends of E_{α} values were
311 interpreted in terms of increase and decrease of the molecular mobility of water hypothesized during
312 the occurrence of these processes.

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315 **Acknowledgements**

316 The authors acknowledge the financial support from FIRB 2012, *Clay nanotubes for designing eco-*
317 *compatible smart materials*, funded by the Italian Ministry of University and Research (Project
318 No.RBFR12ETL5) and Dr. Giuseppe Lazzara and Prof. Peter Simon for helpful discussions.

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448 **Table 1** Comparison of experimental DTG peak temperatures T_p and
 449 mass loss percentages for all the steps evaluated from the TG curves
 450 performed at $10\text{ }^\circ\text{C min}^{-1}$ under a stream of air for all the investigated
 451 materials

Process	$T_p / ^\circ\text{C}$			Mass Loss	
	HNTs	etched HNTs	KAO	HNTs	etched
Dehydration (step 1)	37.5	39.5	n.d.	2.3	4.
Dehydration (step 2)	245	n.d.	n.d.	3.3	n.d.
Dehydroxylation	468	479	503	11.4	6.
Thermal Decomposition (loss of sulfide)	744	730	n.d.	1.7	3.

452 n.d. = not detected

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454

455 **Captions of the figures**

456 **Fig. 1** *a* STEM image of pristine HNTs, *b* STEM image of etched HNTs, *c* SEM image of pristine
 457 HNTs, *d* SEM image of etched HNTs.

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459 **Fig. 2** *a* TG and *b* DTG curves of the materials investigated under a stream of air at $10\text{ }^\circ\text{C min}^{-1}$.

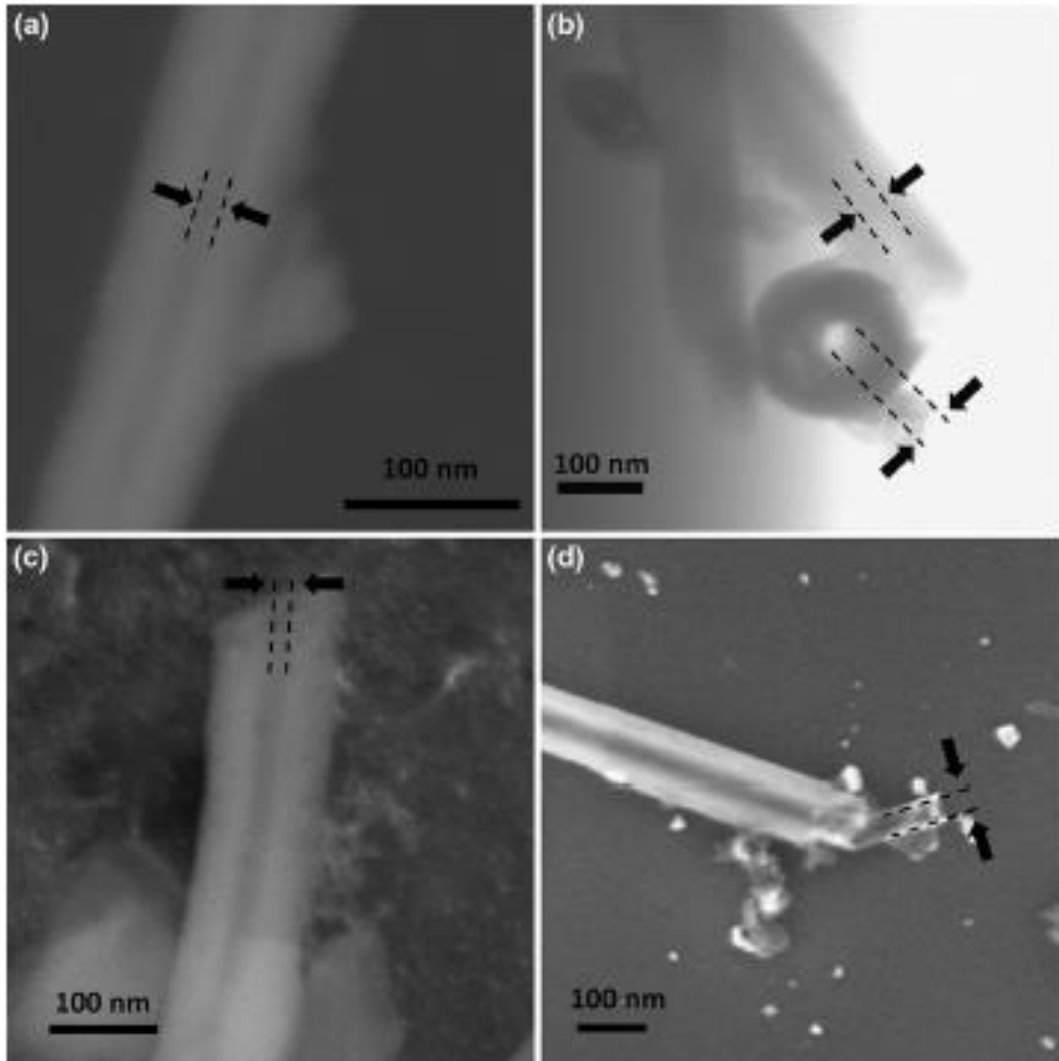
460 **Fig. 3** FTIR spectra of gases evolved during the TG experiments at *a* 39, 245 and 468°C , *b* and
 461 750°C .

462 **Fig. 4** Isoconversional dependency of activation energy of dehydration (loss of layered water
 463 molecules) occurring in pristine HNTs according to the different kinetic methods

464 **Fig. 5** Isoconversional dependencies of activation energy of dehydroxylation (condensation of water
 465 due to dehydroxylation of hydroxyl groups of alumina inner sheets) according to the different
 466 kinetic methods for *a* HNTs, *b* etched HNTs, *c* KAO.

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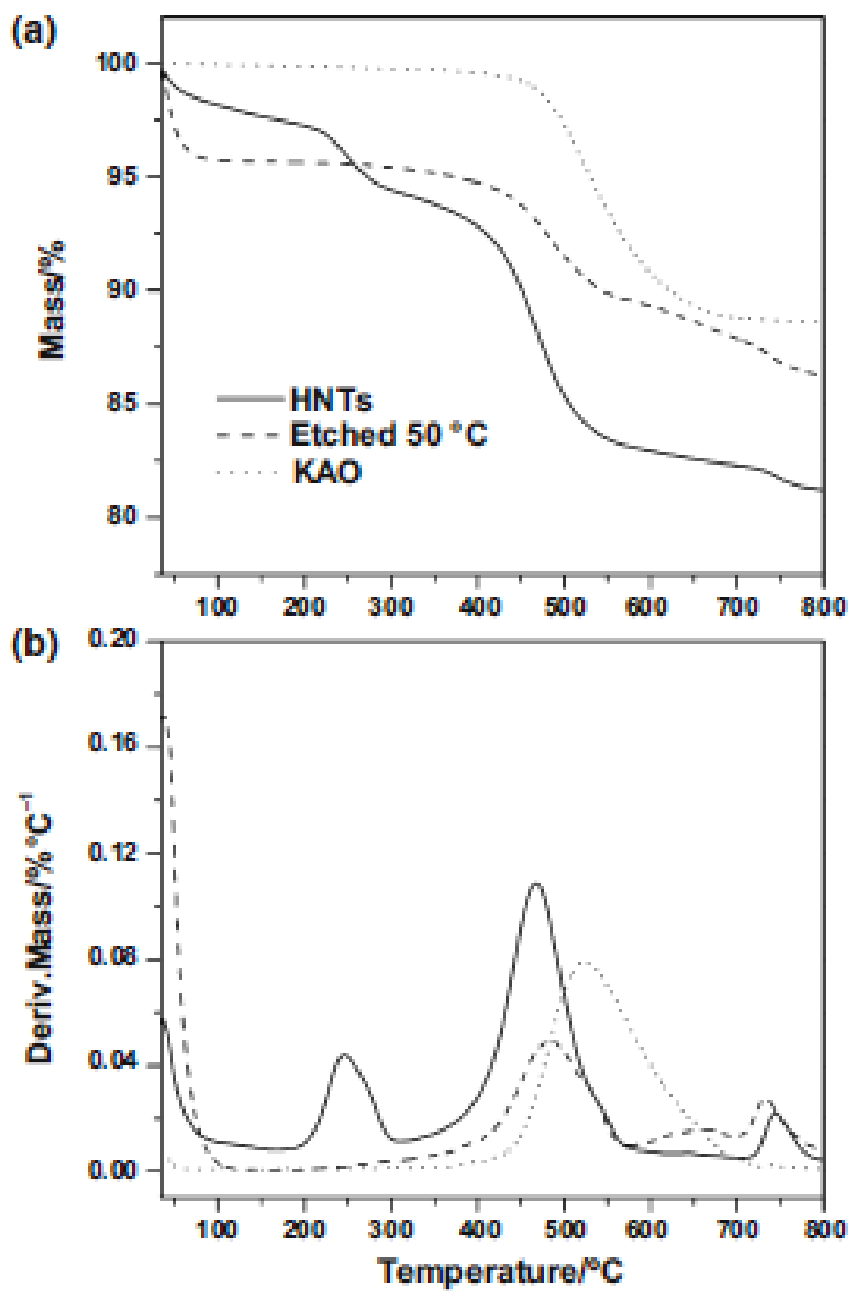
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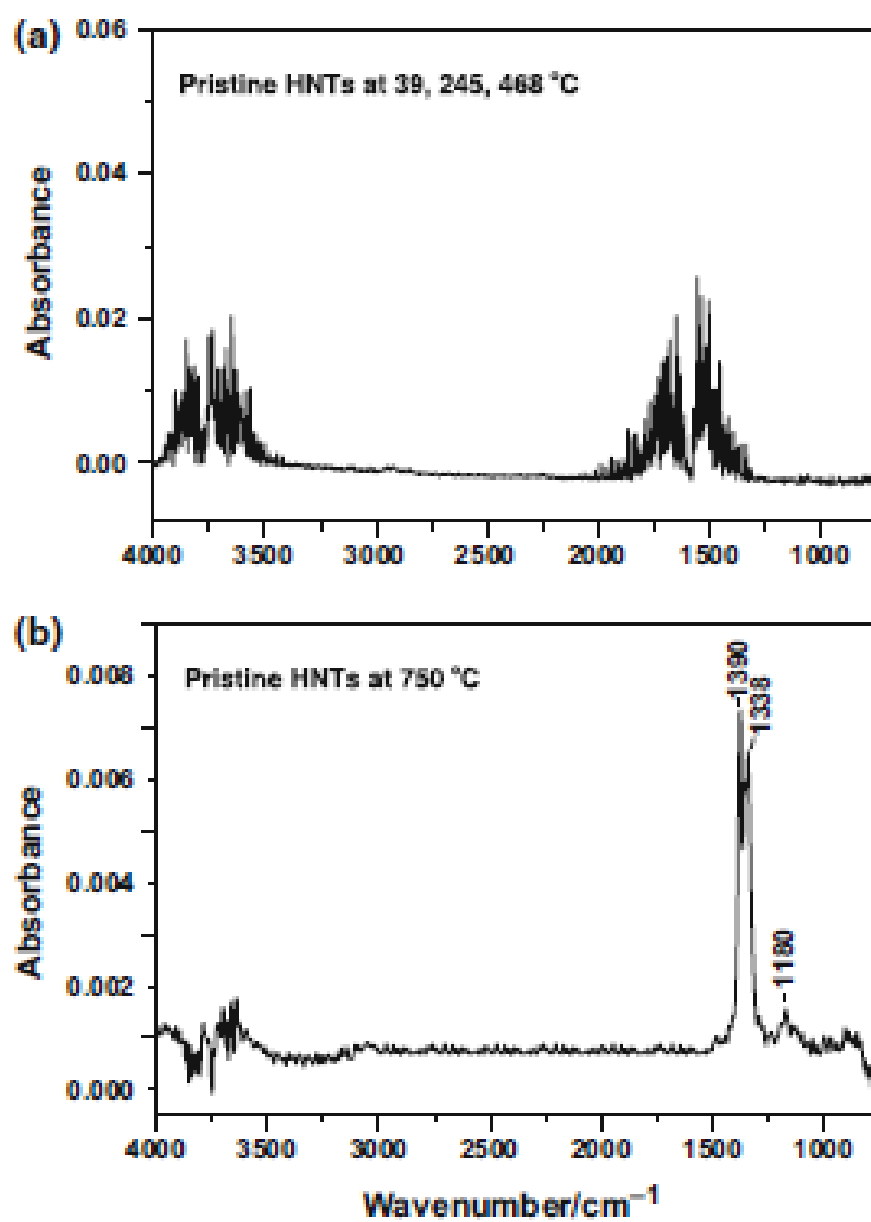


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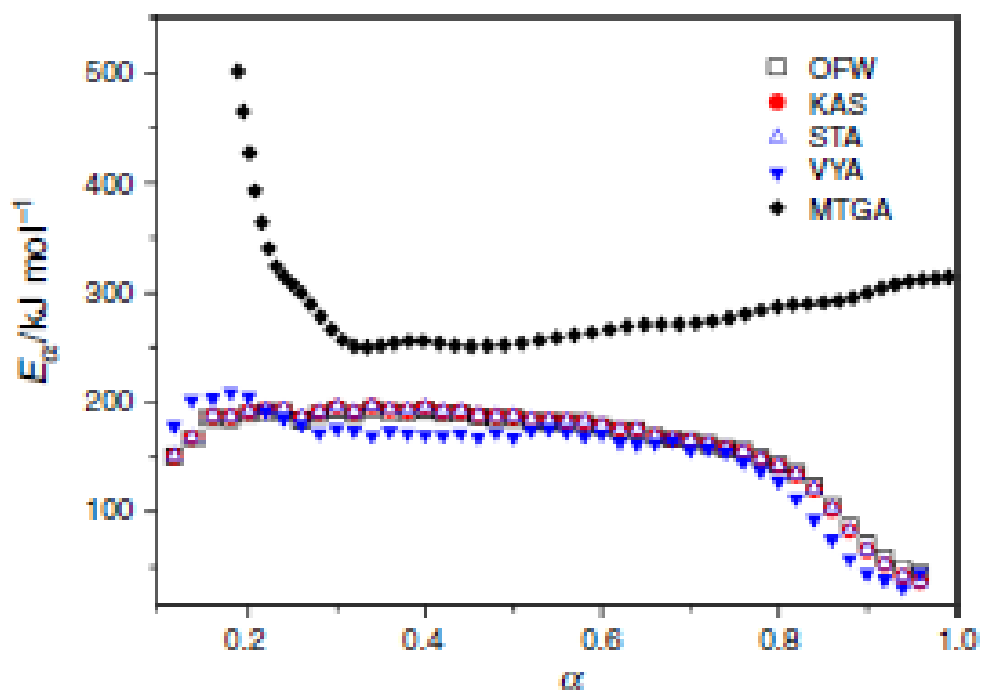
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