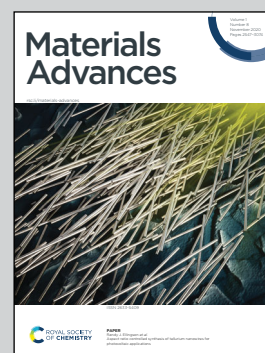


**Showcasing research from Professor Andrea G. Marrani's laboratory, Department of Chemistry, University of Rome "La Sapienza", Rome, Italy.**

A comparative experimental and theoretical study of the mechanism of graphene oxide mild reduction by ascorbic acid and *N*-acetyl cysteine for biomedical applications

Use of green reductants to produce graphene from graphene oxide (GO) results in a biocompatible, low-cost, and environmentally friendly chemical route. In the present work, we explored in parallel the behaviour of *N*-acetyl cysteine (NAC) and ascorbic acid ( $H_2A$ ) towards GO mild reduction. For the first time, we shed light on the reaction mechanisms of GO mild reduction, through electrochemical, XPS, Raman and cell viability experiments, coupled with DFT investigation. Suitable biological tests enlightened the advantages offered by NAC compared to  $H_2A$ , demonstrating the potential benefits in biomedical advanced applications.

**As featured in:**



See Andrea Giacomo Marrani *et al.*,  
*Mater. Adv.*, 2020, 1, 2745.

## PAPER



Cite this: *Mater. Adv.*, 2020,  
1, 2745

# A comparative experimental and theoretical study of the mechanism of graphene oxide mild reduction by ascorbic acid and *N*-acetyl cysteine for biomedical applications†

Andrea Giacomo Marrani,<sup>a</sup> Alessandro Motta,<sup>ab</sup> Valentina Palmieri,<sup>cd</sup> Giordano Perini,<sup>c</sup> Massimiliano Papi,<sup>id</sup><sup>c</sup> Enrique A. Dalchiele,<sup>e</sup> Ricardo Schreiber<sup>f</sup> and Robertino Zanoni<sup>id</sup><sup>a</sup>

A first comparison of the behavior of *N*-acetyl cysteine (NAC) and ascorbic acid (H<sub>2</sub>A) towards reduction of graphene oxide (GO) is reported, along with the novel proposition of the associated reaction mechanisms. NAC and H<sub>2</sub>A are green multi-valent reducing agents, which lead to a mild and biocompatible chemical reduction of oxygenated functional groups in GO. Such reduction has been demonstrated to significantly improve the suitability of GO as a substrate in biomedical applications. A sequence of electrochemical and spectroscopic experiments, theoretical computations and biological tests has been applied to two related series of GO samples mildly reduced with NAC and H<sub>2</sub>A. These display a downshift of the electrochemical reduction potential characteristic of epoxy and carbonyl functional groups, associated with an increase in the electron affinity of the substrates. This potential shift, in turn, makes visible a not previously reported reduction feature, associated with hydroxyl groups. Theoretical modelling unveils the reduction mechanisms operating for H<sub>2</sub>A and NAC on the GO surface, showing their similarity, but also evidencing that NAC remains permanently bonded to the GO surface after reduction, altering the overall reactivity of the reduced GO samples. The mild reduction effect exerted by NAC is stronger in generating attractive biochemical characteristics on GO when compared to H<sub>2</sub>A. In fact, while administration of NAC to eukaryotic cell lines does not affect the cell viability, this is instead reduced in the case of H<sub>2</sub>A. The resulting mildly reduced GO materials represent a new step in the direction of GO derivatives with tailored functionalities and oxidation degree for optimized biomedical applications.

Received 26th June 2020,  
Accepted 30th August 2020

DOI: 10.1039/d0ma00456a

rsc.li/materials-advances

## Introduction

The use of green reductants for the production of various forms of graphene has been widely reported in the last ten years mainly because it couples a biocompatible chemical route starting from graphene oxide (GO) with low-cost and environmentally friendly production of large-area reduced graphene

oxide (rGO). The earliest results were reported with the use of *L*-ascorbic acid on GO<sup>1</sup> and by reducing sugars, such as glucose, fructose and sucrose, on graphene nanosheets.<sup>2</sup> These early reports were followed by the use of a number of different green chemicals applied using distinct approaches, as can be found in various reviews appearing in the last few years.<sup>3–5</sup>

In a recent communication, we first proposed a new green method to produce mildly reduced GO (m-rGO), which is based on the use of *N*-acetyl cysteine (NAC) for different incubation times, as demonstrated by a series of spectroscopic (XPS, Raman, UV-vis), morphological (DLS, AFM, z-potential) and biological results.<sup>6</sup> We pioneered the use of NAC in an effort to obtain additional properties of the resulting m-rGO material, which displayed the persistence of NAC at the surface, where it can act as a radical scavenger.<sup>6</sup>

As reported above, *L*-ascorbic acid (H<sub>2</sub>A) has been earlier and widely investigated as a green GO reductant,<sup>7,8</sup> but its corresponding mechanism of GO reduction did not receive due attention, apart from tentative hypotheses in two previous

<sup>a</sup> Dipartimento di Chimica, Università degli Studi di Roma "La Sapienza", Piazzale Aldo Moro 5, I-00185, Rome, Italy. E-mail: andrea.marrani@uniroma1.it

<sup>b</sup> INSTM UdR Roma, Piazzale Aldo Moro 5, I-00185, Rome, Italy

<sup>c</sup> Department of Neuroscience, Università Cattolica del Sacro Cuore, Largo Francesco Vito, 1, 00168 Rome, Italy

<sup>d</sup> Fondazione Policlinico Universitario A. Gemelli IRCCS, Largo Francesco Vito, 1, 00168 Rome, Italy

<sup>e</sup> Instituto de Física & CINQUIFIMA, Facultad de Ingeniería, Julio Herrera y Reissig 565, C.C. 30, 11000 Montevideo, Uruguay

<sup>f</sup> Instituto de Química, Facultad de Ciencias, Pontificia Universidad Católica de Valparaíso, Av. Brasil, 2950, 2340000 Valparaíso, Chile

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0ma00456a

reports.<sup>3,7</sup> A comparison between H<sub>2</sub>A and NAC is important and timely, particularly in the light of their distinct behaviour towards GO reduction.

In the present work, we explored in parallel the overall behaviour of NAC and H<sub>2</sub>A towards GO mild reduction. According to the incubation time, a series of samples has been investigated, which are referred to as NAC 4 h, NAC 24 h, and NAC 72 h and H<sub>2</sub>A 4 h, H<sub>2</sub>A 24 h, and H<sub>2</sub>A 48 h. These samples underwent detailed electrochemical and spectroscopic characterization plus DFT based theoretical modelling of the reduction reaction mechanisms. The resulting behaviours were compared *via* analysis of cell viability, in order to fully and critically explore the advantages of using mild green reductants in biomedicine.

## Experimental

### General

All the chemicals used here were of analytical reagent grade and were used as received. Doubly distilled water (ddH<sub>2</sub>O) was used.

### Synthesis of GO and mildly chemically reduced GO

A graphene oxide (GO) water dispersion (4 mg mL<sup>-1</sup>) was purchased from Graphenea (Graphenea Inc., Cambridge, MA, USA) and kept in the dark at room temperature till usage. This GO was synthesized using a modified Hummers oxidation method as indicated by the supplier. The control GO samples were prepared by sonicating the pristine commercial product for 30 min and then diluting it with distilled water down to a concentration of 0.1 mg mL<sup>-1</sup>. For GO reduction in L-ascorbic acid (Sigma Aldrich, USA) 900 μL of GO (0.1 mg mL<sup>-1</sup>) was mixed with 100 μL of L-ascorbic acid (H<sub>2</sub>A) solution in water at a concentration of 10 mg mL<sup>-1</sup>, for a final concentration of H<sub>2</sub>A of 1 mg mL<sup>-1</sup> and of GO of 0.09 mg mL<sup>-1</sup>, as reported previously.<sup>6</sup> Reduction (partial and complete) was obtained keeping the solution at room temperature for time intervals between 4 hours and 48 hours. At fixed time points (4, 24 and 48 hours) the reaction was stopped by eliminating H<sub>2</sub>A from the suspension with two centrifugation steps at 14 000 rpm for 10 minutes. At each step, the pellet was re-suspended in fresh ddH<sub>2</sub>O. The finally obtained mildly reduced GO (m-rGO) solutions are denominated H<sub>2</sub>A 4 h, H<sub>2</sub>A 24 h and H<sub>2</sub>A 48 h. For the reduction of GO with N-acetyl-L-cysteine (Calbiochem, USA), 100 μL of a solution of 400 mM N-acetyl-L-cysteine (NAC) was mixed with 900 μL of GO solution (0.1 mg mL<sup>-1</sup>) and then the reduction was performed as described above for H<sub>2</sub>A, with the only difference being that the incubation time ranged from 4 to 72 hours. Therefore, the m-rGO solutions obtained after reduction with NAC are denominated NAC 4 h, NAC 24 h and NAC 72 h. In all cases, after reduction each sample was sonicated for 5 minutes to obtain homogeneous solutions. The hydrodynamic radius of the samples was assessed by Dynamic Light Scattering (DLS) as previously reported.<sup>6</sup> All samples displayed a size below 250 nm after sonication.

### Electrochemical study of GO and m-rGO samples

First, GO and m-rGO layers were deposited onto glassy carbon electrodes (GCE), which were pre-treated as follows. Before the

modification, the GCE was sequentially polished with 0.3 and 0.05 μm Al<sub>2</sub>O<sub>3</sub> powder and then washed successively with anhydrous ethanol and ddH<sub>2</sub>O in an ultrasonic bath and dried in air.

GO and m-rGO thin films were prepared by casting a 50 μL drop of GO and m-rGO suspensions, previously sonicated for 30 min, onto the GCE surface (diameter of 3 mm), and dried naturally at room temperature in air.

All the electrochemical experiments were carried out in a conventional three electrode glass cell comprised of a working electrode (GO or m-rGO film over GCE), a counter electrode (Pt wire) and a reference electrode (Ag/AgCl (KCl 3.5 M),  $U = 0.205$  V vs. NHE). All the potentials in this study are given with reference to this reference electrode. Electrochemical experiments were performed in 0.1 M phosphate (K<sub>2</sub>HPO<sub>4</sub>/KH<sub>2</sub>PO<sub>4</sub>) buffered saline (PBS) solution as a supporting electrolyte; the pH was fixed at 7.2. The electrochemical reduction of the GO and m-rGO samples was carried out by cyclic voltammetry (CV) within the potential window 0.2 to -1.6 V, at a 20 mV s<sup>-1</sup> potential scan rate. Electrochemical measurements were performed using a Bio-Logic SP-150 potentiostat/galvanostat driven by the Bio-Logic EC-Lab<sup>®</sup> software.

### X-ray photoelectron spectroscopy characterization

In order to perform XPS measurements, GO and m-rGO suspension samples were deposited onto Si(111) surfaces, pre-treated as reported elsewhere.<sup>9</sup>

GO and m-rGO thin films were prepared by casting a 50 μL drop of the corresponding suspension, sonicated for 30 min immediately before use, onto the surface of H-terminated Si(111) substrates, and dried in air under heat (40 °C) for 15 min.

XPS measurements were carried out using a modified Omicron NanoTechnology MXPS system equipped with a monochromatic Al Kα ( $h\nu = 1486.7$  eV) X-ray source (Omicron XM-1000), operating the anode at 14 kV and 16 mA. All the photoionization regions were acquired using an analyser pass energy of 20 eV, and take-off angles of 21° with respect to the sample surface normal were adopted. The experimental spectra were theoretically reconstructed by fitting the secondary electron background to a Shirley function and the elastic peaks to pseudo-Voigt functions described by a common set of parameters (position, FWHM, Gaussian-Lorentzian ratio) free to vary within narrow limits. The accuracy of the binding energy values was ±0.05 eV, while the error associated with quantitation was ±10%.

### Raman spectroscopy characterization

Raman spectra of the m-rGO films deposited onto Si wafers were acquired at room temperature in backscattering geometry with an inVia Renishaw micro-Raman spectrometer equipped with an air-cooled CCD detector and super-Notch filters. The emission line at 488.0 nm from an Ar<sup>+</sup> ion laser was focused on the sample under a Leica DLML microscope using a 5× objective. The power of the incident beam was about 5 mW. The spectra were calibrated using the 520.5 cm<sup>-1</sup> line of

the silicon wafer used as a support for the m-rGO deposits, and 10 s accumulations were used for each spectrum with a resolution of  $2 \text{ cm}^{-1}$ .

### UV-vis spectroscopy analysis

Optical UV-vis spectra of GO or m-rGO were obtained using a Cytation 3 (Biotek, USA); at least 3 different measurements per sample were averaged. The UV-vis spectra were used to obtain Tauc plots in order to determine the optical bandgap in selected samples.

### Cell viability and evaluation of reactive oxygen species (ROS) production

For cell viability measurements, U87 human glioblastoma cells were seeded on 24-well plates at a concentration of  $1 \times 10^5$  cells  $\text{mL}^{-1}$ . After 24 hours of incubation at  $37^\circ\text{C}$ , 5%  $\text{CO}_2$ , the cells were treated with GO or m-rGO solutions at a final concentration of  $20 \mu\text{g mL}^{-1}$  for 24 hours. After the treatment, the medium was carefully washed and replaced with fresh medium containing  $20 \mu\text{L}$  of CellTiter-Blue (Promega) per  $100 \mu\text{L}$  of medium and incubated in the dark at  $37^\circ\text{C}$  in 5%  $\text{CO}_2$  for 2 hours. The fluorescence intensity was then recorded with a Cytation 3 Cell Imaging Multi-Mode Reader by exciting at  $550 \text{ nm}$  and reading the emission at  $600 \text{ nm}$ .

For the detection of ROS, the fluorinated derivative of 2',7'-di-chlorofluorescein ( $\text{H}_2\text{DCFDA}$ ) was employed. This probe is non-fluorescent until the acetate groups are removed by intracellular esterases and oxidation occurs within cells. Thus, oxidation can be detected by monitoring the increase in fluorescence intensity. This technique allows for the screening of a larger number of cells compared to fluorescence microscopy. Cells were seeded on 24-well plates at a concentration of  $1 \times 10^5$  cells  $\text{mL}^{-1}$ . After 24 hours of incubation at  $37^\circ\text{C}$ , 5%  $\text{CO}_2$ , the cells were treated with GO or m-rGO at a concentration of  $20 \mu\text{g mL}^{-1}$  for 24 hours. After the treatment, the medium was carefully washed and replaced with PBS containing  $10 \mu\text{M}$   $\text{H}_2\text{DCFDA}$ . The cells were incubated for an additional hour at  $37^\circ\text{C}$ , in 5%  $\text{CO}_2$ . PBS containing  $\text{H}_2\text{DCFDA}$  was then removed and the cells were resuspended in complete medium. The fluorescence intensity of  $\text{H}_2\text{DCFDA}$  was measured by using a Cytation 3 Cell Imaging Multi-Mode Reader by exciting at  $495 \text{ nm}$  and recording the emission at  $528 \text{ nm}$ .

### Computational details

Calculations were performed adopting the B3LYP hybrid GGA functional. The standard all-electron 6-31G\*\* basis<sup>10</sup> was used for all atoms. Molecular geometry optimization of stationary points was carried out without symmetry constraints, by analytical gradient techniques. Frequency analysis was performed to obtain thermochemical information about the reaction pathways at  $298 \text{ K}$  and  $1 \text{ atm}$  using the harmonic approximation. All calculations were performed using the G16 code<sup>11</sup> on Linux cluster systems. Graphene and graphene oxide were modelled adopting a cluster approach. For the reduction process only basal epoxy and edge carbonyl groups were considered.

## Results

The two series of samples produced by contacting GO flakes with NAC or  $\text{H}_2\text{A}$  solutions for different extents of time at room temperature were subjected to a sequence of electrochemical and spectroscopic measurements, and to biological tests. The corresponding chemical reduction mechanisms were investigated with theoretical computations as well. The results are reported and discussed in the following sections.

### Electrochemical study and LUMO levels

Fig. 1 reports the results from CV measurements made on the two series of m-rGO/GCE electrode samples denoted as NAC 4 h, 24 h, and 72 h and  $\text{H}_2\text{A}$  4 h, 24 h and 48 h, as well as the CV of the control GO/GCE one.

In all the current/potential ( $U$ ) voltammograms, an irreversible cathodic feature is visible in the range  $-0.8$  to  $-1.35 \text{ V}$ , with intensities depending on the samples. As we already reported in the case of GO films, this peak can be associated with the electrochemical reduction of epoxy and carbonyl groups, respectively bound to the basal plane or at the edges of GO.<sup>9,12</sup> After the first voltammetric cycle no significant reduction signal can be detected (see Fig. S1–S7 in the ESI†), suggesting that the reduction of these electroactive groups is quantitative within the first potential scan. The charge exchanged during the reduction process, determined by the area of the main CV peak (see Fig. 1 and Table S1 in the ESI†), decreases along with the reduction time in both the NAC and  $\text{H}_2\text{A}$  samples. The current decay trend is strictly related to the abundance of epoxy and carbonyl functional groups on the GO surface.

Another relevant feature evident in the cyclic voltammograms of the m-rGO samples is a further reduction signal

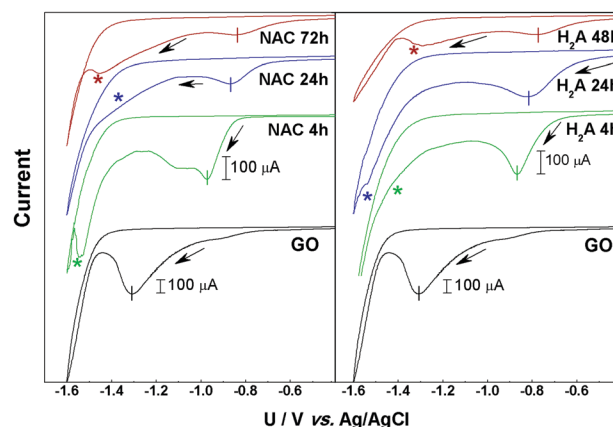


Fig. 1 Left panel: Cyclic voltammograms of NAC 4 h (green), NAC 24 h (blue), and NAC 72 h (red) samples drop cast onto GCE. Right panel: Cyclic voltammograms of  $\text{H}_2\text{A}$  4 h (green),  $\text{H}_2\text{A}$  24 h (blue), and  $\text{H}_2\text{A}$  48 h (red) samples drop cast onto GCE. Measurements were run in  $0.1 \text{ M}$   $\text{pH} = 7.2$  PBS buffer solution at a  $20 \text{ mV s}^{-1}$  potential scan rate. The cyclic voltammogram of a reference GO/GCE electrode is displayed in black. The current scale has been contracted for this latter sample compared to the m-rGO samples for the sake of clarity (see the scale bars). The arrows indicate the initial potential scan direction.

falling in the potential range  $-1.2$  to  $-1.55$  V (indicated by asterisks in Fig. 1), *i.e.*, immediately before the cathodic current rise due to  $H_2$  evolution. This feature is clearly evident in the NAC 4 h, NAC 72 h,  $H_2A$  24 h, and  $H_2A$  48 h samples, and it is still present as a bump in NAC 24 h and  $H_2A$  4 h, but it is definitely absent in pristine GO.

In the sequence of m-rGO samples differently treated with NAC and  $H_2A$ , the onset potential of the first main CV reduction peak progressively moves toward more anodic potentials as the incubation time with the reductant increases (see Fig. 1). This can be ascribed to an increase of the electron affinity values along the series of m-rGO samples, when compared to GO, as demonstrated here below.<sup>13</sup> Among other relevant parameters (Fermi level, work function, *etc.*), the electron affinity is a key ingredient that defines the electronic structure of all interfaces between a material and the environment, and controls the processes of charge exchange and transport across interfaces in all electronic devices.<sup>14–16</sup> To estimate the electron affinity from the measured redox potentials, it is necessary to correlate the electrochemical potentials to the vacuum level.<sup>16–18</sup> On the basis of the values of the reduction onset potentials ( $U_{\text{onset vs. Ag/AgCl}}$ , see Table S1 in the ESI<sup>†</sup>), the lowest unoccupied molecular orbital (LUMO) energy levels ( $E_{\text{LUMO}}$ ) for the NAC and  $H_2A$  m-rGO samples were calculated. The LUMO energy can be determined through the following two equations:<sup>16,17,19,20</sup>

$$E_{\text{LUMO}} = -e(U_{\text{onset vs. NHE}} + 4.75)$$

$$U_{\text{onset vs. NHE}} = U_{\text{onset vs. Ag/AgCl}} + 0.205$$

The LUMO positions for both the GO and m-rGO samples are shown in Fig. 2 as a function of the chemical reduction time with  $H_2A$  and NAC (see Table S1 in the ESI<sup>†</sup>). It can be seen that upon reduction the LUMO position approaches the Fermi level of graphene,  $-4.5$  eV,<sup>13,21–24</sup> apparently more rapidly in the case of reduction exerted by  $H_2A$ . As each electrode voltammogram only exhibited a main cathodic peak (see Fig. 1), the highest occupied molecular orbital (HOMO) energy levels ( $E_{\text{HOMO}}$ ) for the NAC and  $H_2A$  m-rGO samples have been calculated through the optical bandgap  $E_g$ .<sup>25</sup> In fact, the HOMO positions have been extracted using the following equation:<sup>13,26</sup>

$$E_{\text{HOMO}} = (E_{\text{LUMO}} - E_g)$$

where the optical bandgap  $E_g$  has been calculated from Tauc plots considering an indirect bandgap for graphene oxide,<sup>6</sup> see Fig. S8 in the ESI<sup>†</sup>. The resulting HOMO–LUMO values along with the optical bandgap values are shown in the inset of Fig. 2. It is evident that the HOMO–LUMO level separation, and thus the bandgap, decreases with increasing chemical reduction time. In particular, the comparison between the two sets of m-rGO samples shows that the use of NAC requires longer times (72 h) to attain effects comparable to those manifested by  $H_2A$  at shorter times (24 and 48 h). These results demonstrate that it is possible to modulate the electronic band

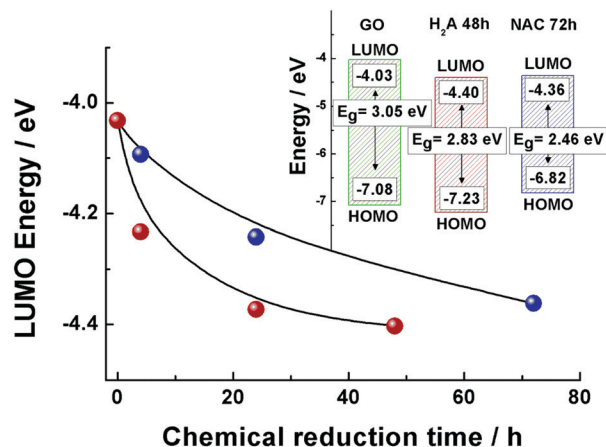


Fig. 2 LUMO energy position estimated from the electrochemical reduction onset potential for m-rGO samples as a function of time of reduction with NAC (blue dots) and  $H_2A$  (red dots). Inset: HOMO position determined from the energy gap ( $E_g$ ) values (estimated from UV-vis absorption spectroscopy) and LUMO (estimated from electrochemical measurements) positions of the NAC 72 h (blue) and  $H_2A$  48 h (red) m-rGO samples compared to the reference GO (green).

parameters by tailoring the surface-oxygenated groups on GO, offering a reliable route to achieve controlled GO band engineering.

### Raman spectroscopy analysis

In order to comparatively probe defects in a related series of analysed samples, Raman spectroscopy represents an ideal tool. A comparison of the Raman spectra of GO and of its electrochemically reduced form, erGO, evidenced the large extent of the reduction characteristic of the latter, as already reported by us.<sup>6,9,27</sup> We undertook Raman analysis in order to investigate in parallel the effect of the two treatment times (24 and 72 h, respectively, for  $H_2A$  and NAC) which gave comparable optical absorbance spectra, LUMO energy levels and similar aggregative states for the rGO suspensions. The results of such parallel analysis between  $H_2A$  and NAC are presented here (Fig. 3), based on the G band, which represents the relative extent of graphitization, and on the so-called “disorder” D and D’ bands, Raman forbidden but activated by defects.<sup>28–30</sup>

Visual inspection of the complex D and G bands for the  $H_2A$  and NAC m-rGO samples reveals that they are modestly changed in the overall shape, if compared to GO.<sup>6</sup> Inspection of the values reported in Table 1 clearly shows that the  $H_2A$  and NAC samples undergo an evolution from GO to a more reduced state upon extension of the incubation time, as inferred from the corresponding values of the  $I_{\text{D}}/(I_{\text{G}} + I_{\text{D}'})$  height ratio. In fact, following the interpretation given in the literature and reported in our previous papers, an increasing trend of this ratio can be considered as a consequence of an increasing quantity of disorder, which is, in turn, associated with the appearance of new carbon  $sp^2$  domains at the expense of carbon–oxygen bonds.<sup>6,9,27,31</sup>

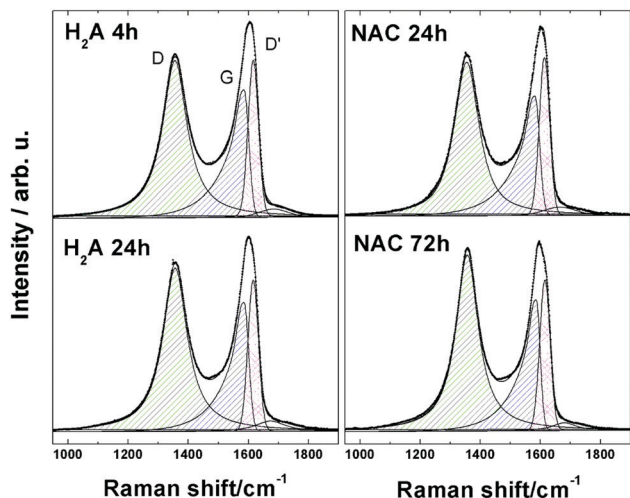


Fig. 3 Raman spectra of m-rGO samples: H<sub>2</sub>A 4 h, H<sub>2</sub>A 24 h, NAC 24 h and NAC 72 h. Experimental data (dots) have been fitted with symmetric and asymmetric Lorentzian model curves (continuous lines).

Table 1 Experimental values of the Raman shift (cm<sup>-1</sup>) and FWHM (cm<sup>-1</sup>) of relevant peaks in the Raman spectra of selected samples addressed in this work. Calculated values of the  $I_D/(I_G + I_{D'})$  height ratio and of the average crystallite size of the sp<sup>2</sup> domains,  $L_a$ , are also reported

Sample	Raman shift (cm <sup>-1</sup> )/FWHM (cm <sup>-1</sup> )			$I_D/(I_G + I_{D'})^a$	$L_a$ (nm)
	D	G	D'		
H <sub>2</sub> A 4 h	1357/99	1586/71	1616/41	0.83	16.4
H <sub>2</sub> A 24 h	1356/98	1586/68	1616/41	0.85	16.0
NAC 24 h	1355/99	1582/73	1614/41	0.85	16.0
NAC 72 h	1357/90	1586/64	1615/41	0.97	13.2
erGO <sup>b</sup>	1356/66	1592/45	1623/30	1.27	10.7
GO <sup>b</sup>	1356/118		1607/86	0.73	18.6

<sup>a</sup> Height ratio. <sup>b</sup> Data taken from ref. 27.

This is further confirmed by the values of the average crystallite size of the sp<sup>2</sup> lattice,  $L_a$ , reported in Table 1 and calculated using the equation:<sup>32</sup>

$$L_a(\text{nm}) = (2.4 \times 10^{-10}) \lambda_{\text{laser}}^4 \left( \frac{I_D}{I_G} \right)^{-1}$$

where  $I_D$  and  $I_G$  are the intensities of the Raman D and G bands, respectively, and  $\lambda_{\text{laser}}$  is the laser wavelength line. These results are all similar and far from what we obtained for erGO, the closer value being represented by NAC 72 h.

The most significant point is that the reduction processes of the H<sub>2</sub>A and NAC samples come to a final extent which is still extra mild, as can be inferred from the trend in FWHM values for peaks D, reported in the literature as a parameter strictly related to the degree of reduction of GO.<sup>33</sup> Such values for the H<sub>2</sub>A and NAC samples fall between those for GO and erGO, but are definitely closer to GO. An analogous trend is consistently followed also by the G and D' peaks, see Table 2, and by the 2D and D + G peaks, see Fig. S9 and Table S2 in the ESI.†

Table 2 Percent atomic ratios from XP spectra<sup>a</sup> of relevant functional groups in the GO and m-rGO samples

Sample	Red.	C-OH	C-O-C	C=O	CO <sub>2</sub> H	C=C <sub>gr</sub> <sup>b</sup>	C=C
GO	—	7.4	30.0	10.8	3.9	—	47.9
m-rGO	NAC 4 h	13.1	20.3	6.4 <sup>c</sup>	7.5 <sup>d</sup>	0.9	51.8
m-rGO	NAC 24 h	10.4	15.4	11.7 <sup>c</sup>	9.7 <sup>d</sup>	7.0	47.8
m-rGO	NAC 72 h	10.3	15.7	11.5 <sup>c</sup>	8.0 <sup>d</sup>	6.9	47.6
m-rGO	H <sub>2</sub> A 4 h	11.5	24.8	8.2	6.5	3.7	45.3
m-rGO	H <sub>2</sub> A 24 h	8.5	24.3	7.9	5.9	6.4	47.0
m-rGO	H <sub>2</sub> A 48 h <sup>e</sup>	13.8	17.9	8.8	11.6	17.3	30.6

<sup>a</sup> Ratios are determined from the area of single components divided by the overall C 1s signal ( $C_{\text{tot}}$ ). The associated error is  $\pm 10\%$ . <sup>b</sup> Component at 284.55 eV in m-rGO samples. <sup>c</sup> Possible presence of C=O from adsorbed NAC. <sup>d</sup> Possible presence of COOH from adsorbed NAC. <sup>e</sup> Possible presence of adsorbed H<sub>2</sub>A.

### X-ray photoelectron spectroscopy characterization

In order to further explore and compare the presence and fate of the various functional groups with the extent of reduction, XPS results coming from the two series of NAC and H<sub>2</sub>A samples are collected in Fig. 4 and discussed in the following.

The C 1s spectra are complex peaks, as expected, and have been subjected to curve fitting in order to extract, assign and

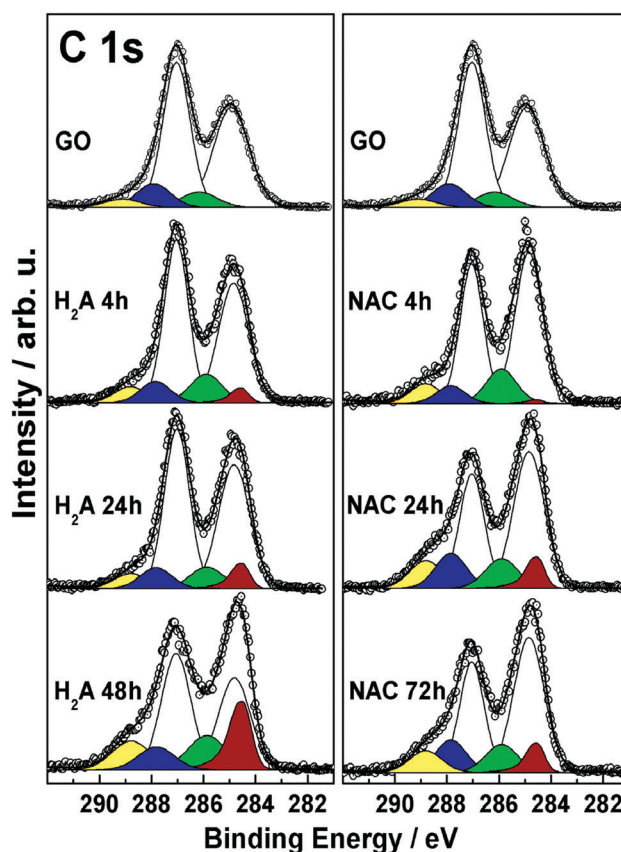


Fig. 4 C 1s XP spectra of the two series of m-rGO samples of this work, with the reference GO spectra reported for comparison. Experimental data are reported as dots, while results of curve-fitting are reported as continuous lines. The distinct components resulting from curve-fitting are coloured as follows: localized C=C (white), delocalized C=C (red), C-OH (green), C-O-C (light cyan), C=O (blue), COOH (yellow).

quantify the corresponding peak components. The sequence of features obtained is consistent with previous reports,<sup>6,9,34,35</sup> and evidences the presence and fate of the distinct C–C and C–O functionalities present at the GO surface at the different reduction steps. In all samples, these functionalities are associated with the following components:  $sp^2$ -hybridized C atoms (284.85 eV, white), and hydroxyl (285.85 eV, green), epoxy (287.05 eV, light cyan), carbonyl (287.85 eV, blue), and carboxyl (288.85 eV, yellow) groups.<sup>36</sup> In addition, in all m-rGO samples a further component is needed to account for the variation in the low-BE line shape after incubation with  $H_2A$  and NAC. This asymmetric feature (284.55 eV, red) is related to the restoration of  $\pi$ -delocalized C=C bonds<sup>6,9,34,35</sup> upon chemical reduction of GO with the green reductants. Upon reduction with NAC, the XP spectra of the resulting m-rGO samples show several differences compared to pristine GO. The most evident is the decrease of the epoxy component (light cyan) with increasing incubation time (see Table 2), mostly occurring in the first 24 h period. At the same time, the graphene-like component (C=C<sub>gr</sub>, red) rises, apparently with the same rate. Other important variations involve the carbonyl and hydroxyl moieties. The former decrease in the short period (NAC 4 h), likely due to consumption after the effect of NAC, but tend to increase again in the long period (see Table 2). Hydroxyl groups, instead, show a discrete but significant increase compared to the parent GO samples.

When  $H_2A$  is used to reduce GO, similar effects can be evidenced through the XP spectra in the C 1s region. In fact, the major impact of  $H_2A$  is on the epoxy moieties, yet with a significant change after 24 h of incubation, strictly paralleled by an increase in the graphene-like component (red). This latter component in the  $H_2A$  48 h sample is found to increase up to 17.3% of the total carbon species (see Table 2), which is in keeping with the above reported rapid decrease of the energy position of the LUMO level. The carbonyl component is quickly reduced to around 8% atomic concentration as a consequence of  $H_2A$  chemical attack. As to the hydroxyl component, it shows an increase compared to GO, with no apparent trend. In conjunction, an increase in the concentration of –COOH groups is detected, especially for the  $H_2A$  48 h sample.

A close correspondence has been found for both series of m-rGO samples between the decreasing concentration trend of the epoxy C 1s XPS component intensity and the charge exchanged during the first reduction event (around  $-0.8$  V vs. Ag/AgCl) recorded in the CV experiments of Fig. 1 (see Table S1 in the ESI†). This parallelism is depicted in Fig. 5 for both samples reduced with NAC and  $H_2A$  green reductants. Though the first peak in the electrochemical reduction of GO has been reported to be a convolution of two closely falling signals<sup>12</sup> related to epoxy and carbonyl groups, in the comparison reported in Fig. 5 the XPS atomic percentage of the latter moiety has been discarded, due to the possible contribution from non-electroactive C=O groups of adsorbed NAC molecules. Nevertheless, the decrease of C–O–C significantly parallels the variation of charge exchanged during CV in both series, and further supports the hypothesis of a chemical attack of NAC and  $H_2A$  towards epoxy and carbonyl groups of GO.

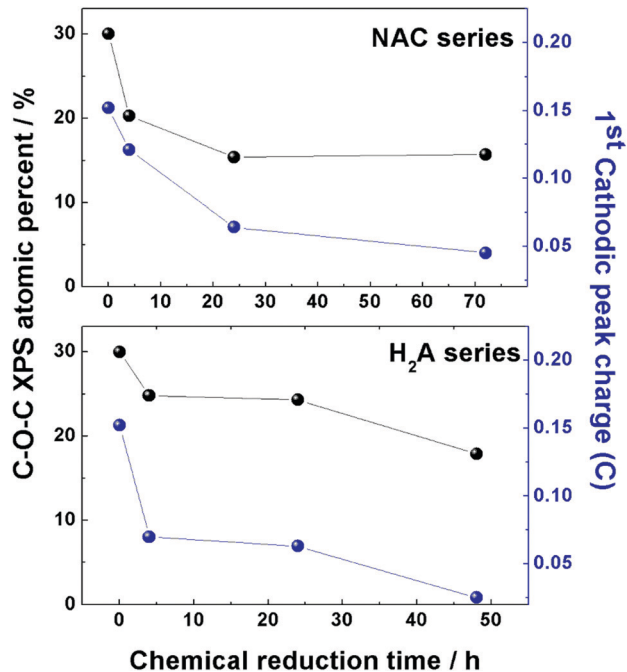


Fig. 5 Comparison between the atomic percentage of epoxy groups (black dots) present in the two series of m-rGO samples, NAC (upper panel) and  $H_2A$  (lower panel), obtained from elaboration of C 1s XP spectra and reported in Table 2, and the charge exchanged (blue dots) during the first reduction event in the CV experiments of the same two series of samples. Data of charge are reported in Table S1 (ESI†).

### Theoretical calculations

Chemical reduction of graphene oxide (GO) was scrutinized by an unprecedented modelling approach of ascorbic acid ( $H_2A$ ) and *N*-acetyl cysteine (NAC) on the GO surface. In both cases, the two prototypical oxygenated functional groups at the GO surface were considered for the reduction process: the basal epoxy and the edge carbonyl groups.

**Ascorbic acid on the GO surface.** When ascorbic acid ( $H_2A$ ) approaches an epoxy group at the GO surface (see Fig. 6a), a hydrogen transfer occurs first between  $H_2A$  and the surface, producing a basal hydroxyl group and a  $HA^\bullet$  radical species.

A fast second hydrogen transfer completes the reduction of the epoxide, producing water and dehydroascorbic acid (DHA), both leaving the surface, locally restoring the graphitic domain.

The overall reaction is highly exoergonic ( $\Delta G = -57.7$  kcal mol<sup>-1</sup>). The first step in Fig. 6a is the rate-determining step, with an energy barrier of 11.2 kcal mol<sup>-1</sup>. The radical intermediate arising from the first H transfer is held by an H-bonding network preparing the correct geometrical arrangement for the second H transfer.

On the other hand, when  $H_2A$  approaches a carbonyl group at the edge of the GO surface, a barrierless exoergonic ( $\Delta G = -9.7$  kcal mol<sup>-1</sup>) H transfer occurs, producing a hydroxyl group bonded to the edge of GO and a radical  $HA^\bullet$  species (see Fig. 6b), which spontaneously leaves the surface ( $\Delta G = -2.5$  kcal mol<sup>-1</sup>). As a further test, the covalent anchoring of  $HA^\bullet$  at the edge of the GO surface was also taken into account. The  $HA^\bullet$  species can

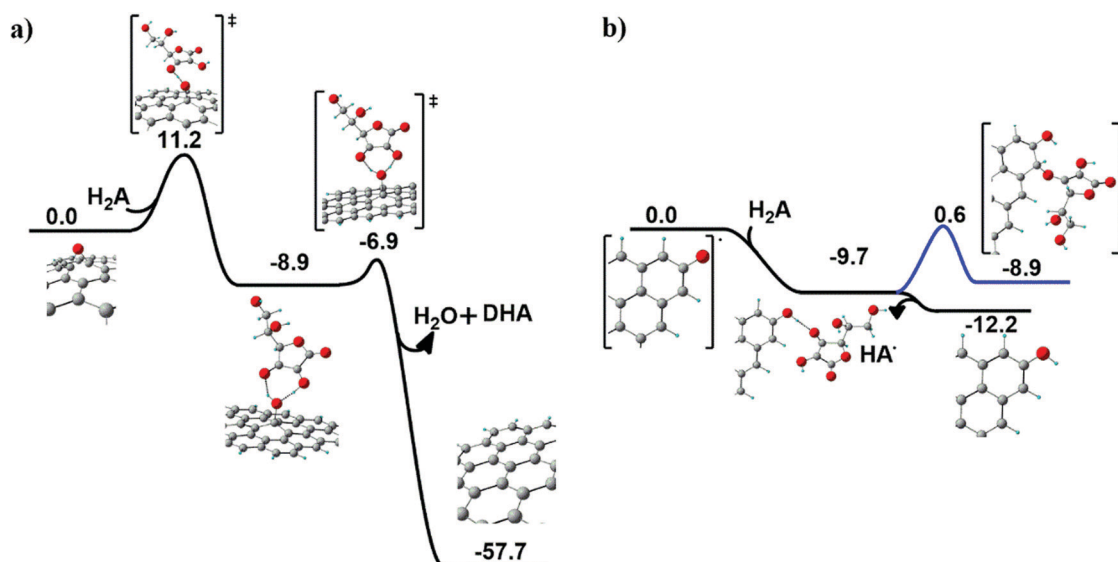


Fig. 6 Computed mechanism of (a) basal epoxy group reduction and (b) carbonyl reduction at the edges of the GO surface induced by H<sub>2</sub>A. Reaction Gibbs free energy (kcal mol<sup>-1</sup>) values are also reported.

coordinate to the carbon vicinal to the hydroxyl group (blue line in Fig. 6b), an isoergonic ( $\Delta G = 0.8$  kcal mol<sup>-1</sup>) process requiring an energy barrier of 10.3 kcal mol<sup>-1</sup> to occur. This step leads to the formation of a new ether group bonded to an sp<sup>3</sup> carbon atom.

Notably, the energy profile depicted in Fig. 6b reveals that the most probable fate associated with the reduction of a GO carbonyl group is the formation of a hydroxyl group in association with the release of HA<sup>•</sup>.

*N*-Acetyl cysteine on the GO surface. When NAC approaches an epoxy group at the GO surface (Fig. 7a), a first hydrogen transfer occurs, producing a basal hydroxyl group and a radical

sulphide species, similarly to the ascorbic acid case. The radical sulphide species resides vicinal to the surface, due to hydrogen bonding between the surface hydroxyl and the acetyl groups. The hydroxyl group is eventually extracted from the surface, forming a sulfenic acid derivative of NAC and restoring the graphitic domain.

Also in this case, the kinetics of epoxide reduction is dominated by the first hydrogen transfer step, with an energy barrier of 22.1 kcal mol<sup>-1</sup>. This value is higher than the energy barrier found in the case of ascorbic acid (11.2 kcal mol<sup>-1</sup>), suggesting a slower (less favourable) process for NAC.

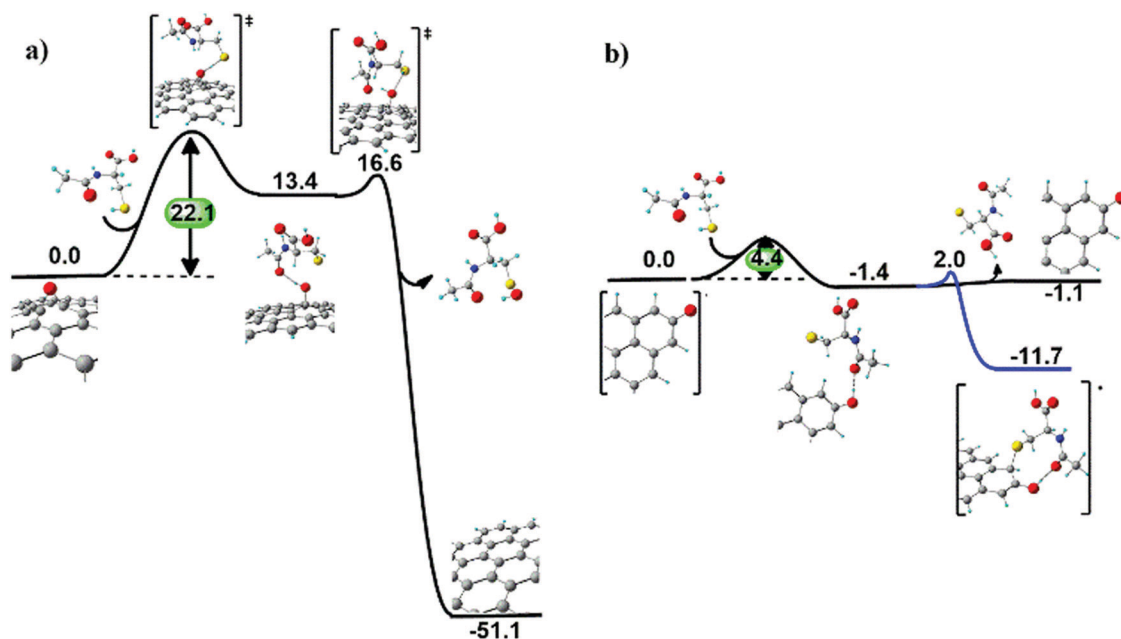


Fig. 7 Computed mechanism of (a) epoxy group reduction and of (b) carbonyl group reduction at the edges of the GO surface induced by NAC. Reaction Gibbs free energy (kcal mol<sup>-1</sup>) values are also reported.



After reduction of a first epoxy group, the resulting sulfenic acid molecule can reduce two more epoxy groups of GO *via* analogous exoergic mechanisms, passing from sulfenic acid and leading up to the inactive sulfonic acid (see Fig. S10 in the ESI†).

Reduction of carbonyl groups at the edges of GO involves a first hydrogen transfer producing a hydroxyl group, as for the ascorbic acid case. However, in this case the covalent bond of the sulphide radical species to the hydroxyl vicinal carbon atom ( $\Delta G = -11.7 \text{ kcal mol}^{-1}$ ) is thermodynamically preferred to the removal of the sulphide radical species ( $\Delta G = -1.1 \text{ kcal mol}^{-1}$ ) from the GO surface (Fig. 7b).

Overall, the three main outcomes of the computed reduction mechanisms of GO exerted by NAC and  $\text{H}_2\text{A}$  are: (i) epoxy groups are fully reduced, with NAC displaying both slightly lower kinetics than  $\text{H}_2\text{A}$  and the capability of reducing three C–O–C groups per molecule; (ii) edge carbonyl groups are partially reduced to hydroxyl groups, and (iii) NAC more favourably binds to rGO (*via* a C–S bond) compared to  $\text{H}_2\text{A}$  (*via* a C–O bond).

### Biological tests

GO has been largely exploited in biomedical applications, from antibacterial formulations to gene therapy and tissue regeneration therapies.<sup>37–39</sup> However, a great concern about this nano-material is its potential cytotoxicity, which must be evaluated prior to the translation of scientific results in clinical settings. Here, a concentration of GO considered safe for *in vitro* administration was used on the same cell line,<sup>40</sup> and we evaluated whether the reduction with  $\text{H}_2\text{A}$  or NAC can change the GO effects on a eukaryotic cell line (U87 human glioblastoma). As already reported in the literature, rGO is more cytotoxic than GO,<sup>40</sup> and in our experiments m-rGO reduced by  $\sim 25\%$  the cell viability (Fig. 8a, asterisk indicates  $p < 0.05$ ) when produced by the  $\text{H}_2\text{A}$  method. Since excess NAC or  $\text{H}_2\text{A}$ , *i.e.* the fraction not adhering to the GO surface, has been removed from the rGO formulations with several centrifugations prior to cell treatment, the cytotoxicity observed can be ascribed to the reduced graphene oxide.

It should be noted that the effect is mild, when compared to the literature, since  $\text{H}_2\text{A}$  is a green reductant. This effect is mediated by the increase of intracellular ROS production, as it is visible from the  $\text{H}_2\text{DCFDA}$  fluorescence increase (Fig. 8b). When NAC is used to perform mild reduction of GO, the viability of cells is restored to levels comparable to GO, and the ROS production is not significantly different from the ROS produced when GO is administered.

## Discussion

The overall picture emerging from the above results shows that the electronic properties of the GO surface can be finely and distinctly tuned by a mild chemical reduction process conducted with  $\text{H}_2\text{A}$  and NAC. The effects of the NAC and  $\text{H}_2\text{A}$  green reductants can be rationalized in terms of partial elimination of epoxy and carbonyl functional groups, which proceeds further in the time frame of hours and days, and renders

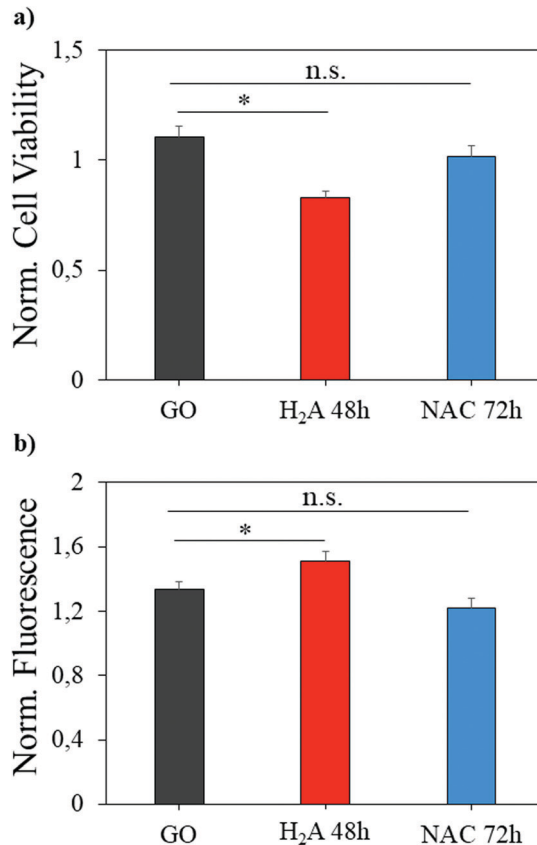


Fig. 8 Cell viability (a) and ROS production (b) induced by administration of different samples to the U87 cell line. Significantly different results are highlighted with asterisks ( $p < 0.05$ ).

more and more facile the electrochemical reduction of the residual oxygenated groups.

The not previously reported CV reduction signal of m-rGO, experimentally found before the  $\text{H}_2$  reduction peak (Fig. 1), suggests that its origin could primarily stem from the modifications induced by the green reductants on the structural electronic properties of GO. In fact, mild chemical reduction induces a partial restoration of the  $\text{sp}^2$  C network, more efficiently produced by the use of  $\text{H}_2\text{A}$  than NAC, as consistently demonstrated by a wide set of experimental data. These include the downshift of the LUMO position associated with an increase in the electron affinity (Fig. 2), together with the increased concentration of  $\text{sp}^2$  domains observed by Raman spectroscopy (Fig. 3) and the trend of the graphitic component observed in the C 1s XPS region (Fig. 4). Such restoration brings together a downshift of all the reduction potentials. The newly reported reduction signal of m-rGO can then be confidently associated with the reduction of the hydroxyl groups on the GO surface, which are expected to undergo electrochemical reduction at potentials more negative than epoxy and carbonyl groups.<sup>41</sup>

The above assignment is further corroborated and refined by theoretical analysis, which contributes to explaining how the mild reducing agents modify the GO surface during the reduction process. In fact, the increasing abundance of hydroxyl groups observed for both the NAC and  $\text{H}_2\text{A}$  reductants

(Table 2) is related to the fate of the edge carbonyls present on the GO surface, as demonstrated by the modelled energetic profiles of the reduction process of C=O groups due to both the H<sub>2</sub>A (Fig. 6) and NAC (Fig. 7) reducing agents. In the case of H<sub>2</sub>A, the reasons for this increase might also be related to the residual presence of H<sub>2</sub>A (and/or its derivative species) adsorbed at the surface of GO. In this regard, the experimentally found increase of the COOH component (Table 2) may as well be contributed by the lactone group of adsorbed H<sub>2</sub>A molecules. As a consequence, the appearance of the reduction signal of m-rGO can be attributed to the increasing hydroxyl group abundance in addition to the downshift of all the reduction current potentials, as previously described.

The distinct modes of operation of GO reduction exerted by H<sub>2</sub>A and NAC are further evidenced by the spontaneous bonding of NAC to the GO edges, suggested by theoretical calculations. This result is in good agreement with the increase in concentration of C=O groups observed by XPS (Table 2) during the reduction process with NAC, and it is related to the permanence of NAC molecules (bearing an acetyl group), or their derivatives, at the surface of m-rGO. In this regard, a recent report from this group showed that NAC persists at the surface of GO, and demonstrated that it exerts prolonged radical scavenging activity in the form of rGO–NAC adducts.<sup>6</sup> As support for this explanation, the carboxyl group signal increases as well, after incubation of GO with NAC (Table 2).

The distinct GO reduction mechanisms evidenced for the two green reductants strongly influence the biological response of the different m-rGO samples. In fact, we have demonstrated that, when NAC-reduced m-rGO samples are administered to eukaryotic cells, the ROS production levels are comparable to those related to GO. Therefore, reduction with NAC leads to rGO materials less cytotoxic than those obtained from H<sub>2</sub>A, which is in full accord with the already reported negligible oxidative capability exerted by NAC on glutathione.<sup>6</sup> This opens up new routes in the possible applications of non-cytotoxic rGO materials for biomedical purposes.

## Conclusions

A novel and thorough comparison has been presented between the distinct GO reduction behaviours of H<sub>2</sub>A, the earliest green reductant applied to GO, and of NAC, very recently applied in such a process. For the first time, the reaction mechanisms related to mild GO reduction have been investigated in detail, by making use of results coming from electrochemical, electron affinity, XPS, Raman and cell viability experiments, coupled with a detailed theoretical investigation of the reactivity of NAC and H<sub>2</sub>A at the GO surface.

GO has been shown to undergo a very mild reduction process by either of the two green chemicals, albeit with distinct behaviours. A change in the electron affinity levels of the mildly reduced substrates has been evidenced, quantified, and employed to explain the first experimental observation of an electrochemical feature related to the presence of hydroxyl groups. Relevant and

subtler differences between NAC and H<sub>2</sub>A have been extracted by the combined approach here reported, which can be beneficial for tailoring GO towards its reproducible use as a substrate. Suitable biological tests, here conducted, have enlightened the advantage offered, in such a process, by NAC with respect to the earlier and widely reported use of H<sub>2</sub>A, which can be particularly beneficial in biomedical advanced applications.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

A. G. M. thanks Sapienza University of Rome for financial support through the Ateneo Projects 2018 (no. RG1181643265D950) and 2019 (no. RM11916B88D8E044). E. A. D. acknowledges CSIC (Comisión Sectorial de Investigación Científica), Universidad de la República, Montevideo, Uruguay, and PEDECIBA – Física. R. S. thanks CONICYT, Chile, through Proyecto FONDECYT No. 1160485, and DII (Pontificia Universidad Católica de Valparaíso, Chile). A. M. acknowledges CINECA award N. HP10CXE6KL 2019 under the ISCRA initiative for the computational resources supporting this work. M. P. and V. P. acknowledge funding from AIRC under IG 2019 – ID. 23124 project. M. P. and R. Z. acknowledge MIUR for funding of PRIN project “ASSEMBLe” 2017RSAFK7.

## References

- 1 M. J. Fernández-Merino, L. Guardia, J. I. Paredes, S. Villar-Rodil, P. Solís-Fernández, A. Martínez-Alonso and J. M. D. Tascón, *J. Phys. Chem. C*, 2010, **114**, 6426–6432.
- 2 C. Zhu, S. Guo, Y. Fang and S. Dong, *ACS Nano*, 2010, **4**, 2429–2437.
- 3 C. Xu, X. Shi, A. Ji, L. Shi, C. Zhou and Y. Cui, *PLoS One*, 2015, **10**, e0144842.
- 4 M. T. H. Aunkor, I. M. Mahbulbul, R. Saidur and H. S. C. Metselaar, *RSC Adv.*, 2016, **6**, 27807–27825.
- 5 K. K. H. De Silva, H. H. Huang, R. K. Joshi and M. Yoshimura, *Carbon*, 2017, **119**, 190–199.
- 6 V. Palmieri, E. A. Dalchiele, G. Perini, A. Motta, M. De Spirito, R. Zanoni, A. G. Marrani and M. Papi, *Chem. Commun.*, 2019, **55**, 4186–4189.
- 7 J. Gao, F. Liu, Y. Liu, N. Ma, Z. Wang and X. Zhang, *Chem. Mater.*, 2010, **22**, 2213–2218.
- 8 J. Zhang, H. Yang, G. Shen, P. Cheng, J. Zhang and S. Guo, *Chem. Commun.*, 2010, **46**, 1112–1114.
- 9 A. G. Marrani, R. Zanoni, R. Schrebler and E. A. Dalchiele, *J. Phys. Chem. C*, 2017, **121**, 5675–5683.
- 10 V. A. Rassolov, J. A. Pople, M. A. Ratner and T. L. Windus, *J. Chem. Phys.*, 1998, **109**, 1223.
- 11 M. J. Frisch, *et al.*, *GAUSSIAN 16, Revision B.01*, Gaussian Inc., Wallingford, CT, 2016.
- 12 A. G. Marrani, A. Motta, R. Schrebler, R. Zanoni and E. A. Dalchiele, *Electrochim. Acta*, 2019, **304**, 231–238.

- 13 F. Zheng, W.-L. Xu, H.-D. Jin, X.-T. Hao and K. P. Ghiggino, *RSC Adv.*, 2015, **5**, 89515–89520.
- 14 A. Kahn, *Mater. Horiz.*, 2016, **3**, 7–10.
- 15 S. Lindsay, *J. Chem. Educ.*, 2005, **82**, 727–733.
- 16 F. N. Crespilho, V. Zucolotto, J. R. Siqueira, A. J. F. Carvalho, F. C. Nart and O. N. Oliveira, *Int. J. Electrochem. Sci.*, 2006, **1**, 151–159.
- 17 C. M. Cardona, W. Li, A. E. Kaifer, D. Stockdale and G. C. Bazan, *Adv. Mater.*, 2011, **23**, 2367–2371.
- 18 B. W. Larson, J. B. Whitaker, X.-B. Wang, A. A. Popov, G. Rumbles, N. Kopidakis, S. H. Strauss and O. V. Boltalina, *J. Phys. Chem. C*, 2013, **117**, 14958–14964.
- 19 D. Gedefaw, M. Tassarolo, M. Bolognesi, M. Prosa, R. Kroon, W. Zhuang, P. Henriksson, K. Bini, E. Wang, M. Muccini, M. Seri and M. R. Andersson, *Beilstein J. Org. Chem.*, 2016, **12**, 1629–1637.
- 20 U. A. Méndez-Romero, S. A. Pérez-García, X. Xu, E. Wang and L. Licea-Jiménez, *Carbon*, 2019, **146**, 491–502.
- 21 M. Lundie, Ž. Šljivančanin and S. Tomić, *J. Mater. Chem. C*, 2015, **3**, 7632–7641.
- 22 G. Colherinhas, E. E. Fileti and V. V. Chaban, *J. Phys. Chem. Lett.*, 2015, **6**, 302–307.
- 23 R. Garg, N. Dutta and N. Choudhury, *Nanomaterials*, 2014, **4**, 267–300.
- 24 M. M. Stylianakis, M. Sygletou, K. Savva, G. Kakavelakis, E. Kymakis and E. Stratakis, *Adv. Opt. Mater.*, 2015, **3**, 658–666.
- 25 L. V. Brownell, K. A. Robins, Y. Jeong, Y. Lee and D. C. Lee, *J. Phys. Chem. C*, 2013, **117**, 25236–25247.
- 26 T. Majumder, K. Debnath, S. Dhar, J. J. L. Hmar and S. P. Mondal, *Energy Technol.*, 2016, **4**, 950–958.
- 27 A. G. Marrani, A. C. Coico, D. Giacco, R. Zanoni, F. A. Scaramuzzo, R. Schrebler, D. Dini, M. Bonomo and E. A. Dalchiale, *Appl. Surf. Sci.*, 2018, **445**, 404–414.
- 28 A. C. Ferrari, J. C. Meyer, V. Scardaci, C. Casiraghi, M. Lazzeri, F. Mauri, S. Piscanec, D. Jiang, K. S. Novoselov, S. Roth and A. K. Geim, *Phys. Rev. Lett.*, 2006, **97**, 187401.
- 29 L. M. Malard, M. A. Pimenta, G. Dresselhaus and M. S. Dresselhaus, *Phys. Rep.*, 2009, **473**, 51–87.
- 30 A. Eckmann, A. Felten, A. Mishchenko, L. Britnell, R. Krupke, K. S. Novoselov and C. Casiraghi, *Nano Lett.*, 2012, **12**, 3925–3930.
- 31 X. Diez-Betriu, S. Alvarez-Garcia, C. Botas, P. Alvarez, J. Sanchez-Marcos, C. Prieto, R. Menendez and A. de Andres, *J. Mater. Chem. C*, 2013, **1**, 6905–6912.
- 32 M. A. Pimenta, G. Dresselhaus, M. S. Dresselhaus, L. G. Cançado, A. Jorio and R. Saito, *Phys. Chem. Chem. Phys.*, 2007, **9**, 1276–1291.
- 33 D. S. Lee, C. Riedl, B. Krauss, K. von Klitzing, U. Starke and J. H. Smet, *Nano Lett.*, 2008, **8**, 4320–4325.
- 34 C.-Y. Lin, C.-E. Cheng, S. Wang, H. W. Shiu, L. Y. Chang, C.-H. Chen, T.-W. Lin, C.-S. Chang and F. S.-S. Chien, *J. Phys. Chem. C*, 2015, **119**, 12910–12915.
- 35 R. Larciprete, P. Lacovig, S. Gardonio, A. Baraldi and S. Lizzit, *J. Phys. Chem. C*, 2012, **116**, 9900–9908.
- 36 M. P. Briggs and D. Seah, *Practical Surface Analysis*, J. Wiley & Sons, Chichester, 2nd edn, 1990.
- 37 V. Palmieri, F. Bugli, M. Cacaci, G. Perini, F. De Maio, G. Delogu, R. Torelli, C. Conti, M. Sanguinetti, M. De Spirito, R. Zanoni and M. Papi, *Nanomedicine*, 2018, **13**, 2867–2879.
- 38 V. Palmieri, M. Barba, L. Di Pietro, S. Gentilini, M. C. Braidotti, C. Ciancico, F. Bugli, G. Ciasca, R. Larciprete, W. Lattanzi, M. Sanguinetti, M. De Spirito, C. Conti and M. Papi, *2D Mater.*, 2018, **5**, 015027.
- 39 R. Di Santo, L. Digiacomo, S. Palchetti, V. Palmieri, G. Perini, D. Pozzi, M. Papi and G. Caracciolo, *Nanoscale*, 2019, **11**, 2733–2741.
- 40 S. Jaworski, E. Sawosz, M. Kutwin, M. Wierzbicki, M. Hinzmann, M. Grodzik, A. Winnicka, L. Lipińska, K. Włodyga and A. Chwalibog, *Int. J. Nanomed.*, 2015, **10**, 1585–1596.
- 41 E. L. K. Chng and M. Pumera, *Chem. – Asian J.*, 2011, **6**, 2899–2901.