

# Synthesis of WO<sub>3</sub> nanopowder using a green surfactant for efficient gas sensing applications

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## ABSTRACT

WO<sub>3</sub> has attracted great attention for chemical gas sensing applications. However, the synthesis of WO<sub>3</sub> is mainly performed by using hazardous organic solvents and corrosive acidic solutions, which can result in environmental and health issues. The development of safe synthesis procedures using green compounds and solvents is in great demand to overcome the aforementioned drawbacks. Therefore, the effect of different eco-friendly chemical compounds on the growth of WO<sub>3</sub> materials has yet to be thoroughly studied. Meanwhile, the fabrication of material at the nanoscale with specific crystalline properties may improve its sensitivity to the target gas. In this work, we report an efficient method for the synthesis of WO<sub>3</sub> nanopowder by using water and a green surfactant such as vitamin C. The growth mechanism of the structure is analyzed considering the nature of solutions and surfactants. The studies indicate that the surfactant has a crucial effect on the formation of nanoparticles. Furthermore, the experimental findings show that the fabricated monoclinic WO<sub>3</sub> material is highly sensitive and selective to low concentrations of acetone. Moreover, the structure exhibits quite stable functionalities at different levels of environmental humidity. Hence, this work may have a considerable impact on the existing techniques for the preparation of WO<sub>3</sub> nanomaterials providing new insights into their green synthesis procedures. In addition, it can significantly affect the development of high-performance gas sensors and other catalytic devices based on WO<sub>3</sub> nanostructures.

## 1. Introduction

Nowadays, green chemistry is highly required to reduce the use and production of hazardous substances. In this respect, green synthesis methods are designed considering twelve basic principles of green chemistry [1–3]. Therein, the fabrication of nanostructured materials without using dangerous organic solvents, corrosive acids (for example, HNO<sub>3</sub>, HCl, or H<sub>2</sub>SO<sub>4</sub>), or non-toxic reagents is one of the most significant challenges to minimizing the application of harmful substances [1, 4–8]. Moreover, particular attention should be paid to the enhancement of the efficiency of chemical processes [3,9].

Intensive studies have been carried out to explore green

technological procedures for preparing semiconductor nanostructures with different shapes and their application in energy, medicine, and sensing [8,10–12]. Especially, the large surface area of nanomaterials and the possibility of tailoring their composition and structure make them very attractive candidates for application in chemical gas sensors. Here, the enhancement of the material surface area will affect its interaction with gaseous compounds, while the compositional and structural modifications can improve its reactivity to specific gaseous or volatile compounds [13–15]. The need for high-performance chemical gas sensors based on semiconductor nanomaterials is justified by several factors, such as indoor and outdoor air quality monitoring, safety in industrial processes, and health status diagnosis [16–19]. Therefore, the

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integration of eco-friendly synthesis procedures into the development of chemiresistive gas sensors has received considerable attention [13,20]. Different chemical approaches have been used for the green synthesis of metal oxide nanomaterials. However, oxide nanostructures have a high tendency to agglomerate, and avoiding this phenomenon is crucial for their preparation [7,21–23].

As an n-type oxide semiconductor,  $\text{WO}_3$  has been employed in various applications owing to its unique physical and chemical characteristics such as high stability, biocompatibility, nontoxicity, and low cost [24–26]. The experimental and theoretical investigations suggest that  $\text{WO}_3$  is an attractive material for its application in chemical gas sensing devices [27–31]. However, the green synthesis of  $\text{WO}_3$  nanostructures is in its infancy, and therefore the effect of different approaches, chemicals, and solvents on structure growth should be examined to identify the optimum methods and materials to be used. Recent achievements in green synthesis procedures show that metal oxide nanostructures can be prepared using different biomolecules like vitamins, proteins, amino acids, organic acids, and polyphenols, which can be extracted from other plants and act as structure-directing agents [7,23,24,32]. Among the fabrication techniques, the precipitation approach provides a flexible and cost-effective tool for producing nanomaterials at relatively low temperatures and controlling their morphology, porosity, and size [33–35].

In this paper, we report a simple, low-cost, and environmentally friendly approach for the fabrication of  $\text{WO}_3$  nanopowder in the aqueous medium using vitamin C as a surfactant. The morphology of the material is studied by field-emission scanning electron microscopy (FESEM). The structure and the chemical composition are analyzed by X-ray diffraction (XRD), Raman, and attenuated total reflectance (ATR) Fourier-transform infrared (IR) spectroscopy. Experimental findings indicate the fabrication of powdered material with individual nanoparticles. Meanwhile, the surfactant nature is crucial for the formation of well-defined particles. Chemical gas sensing investigations show that the material has a good and selective response to acetone molecules. Furthermore, the monoclinic crystal structure of prepared  $\text{WO}_3$  significantly affects its sensing performance. Thus, the results of this research study can be employed in green technologies for the synthesis of  $\text{WO}_3$  nanomaterials, as well as in the fabrication of gas sensors and other catalytic systems.

## 2. Experimental

### 2.1. Synthesis of powders

$\text{WO}_3$  nanopowders were synthesized by the precipitation method in different solutions, where tungsten (VI) chloride ( $\text{WCl}_6$ , Sigma-Aldrich) was used as the precursor material. Vitamin C was obtained from Sigma-Aldrich. Potassium sulfate was obtained from Fluka. The solutions were prepared in distilled water (Carlo Erba Reagents) with three different compositions.

- (i) 1 g of  $\text{WCl}_6$  was dissolved in 50 ml of distilled water ( $\text{H}_2\text{O}$ ) and stirred for 5 h at room temperature.
- (ii) Aqueous solution of vitamin C was prepared as follows: 1 M aqueous solution of vitamin C was stirred for 20 min at room temperature. Then,  $\text{WCl}_6$  was dissolved in 50 ml of  $\text{H}_2\text{O}$  and stirred for 30 min at room temperature. Afterward, 1 M solution of vitamin C was added dropwise into the precursor solution ( $\text{WCl}_6/\text{H}_2\text{O}$ ) until the pH value reached 2 and stirred for 5 and 24 h. Herein, vitamin C was used as the green capping, reducing, and pH control agent.
- (iii) The third solution was prepared as the second one, however, instead of 1 M vitamin C, 0.8 M potassium sulfate was added to the solution as the structure-directing agent.

After the preparation procedure, the resulting solutions were kept at

room temperature for 2 h allowing them to settle out. The precipitates were collected by centrifugation. Then, the salt ions as well as the other side-products were removed from precipitates by washing them with  $\text{H}_2\text{O}$  and ethanol. The final powders were dried at room temperature for 24 h and then annealed at 450 °C for 2 h.

### 2.2. Characterizations

The morphological properties of fabricated materials were investigated using high-resolution FESEM (MIRA3 FEG-SEM, TESCAN). The structural properties were analyzed by XRD (Empyrean model, PANalytical, The Netherlands) with a Cu-LFF ( $\lambda = 1.54 \text{ \AA}$ ) source at 40 kV and 40 mA. The  $2\theta$  (Bragg's diffraction angles) measurements were carried out from 15° to 90° with a scan speed of 100 s and a step size of 0.013°.

Raman spectra were recorded by a Jasco NRS-5100 confocal Raman microscope equipped with the excitation laser diode (wavelength: 785 nm, grating: 600 l/mm) that operates at a nominal power of 20 mW. An MPLFLN 100x objective focuses the laser beam on the sample with a 1  $\mu\text{m}$  spot and the backscattered signal is collected using a cooled (−69 °C) CCD camera. The studies were carried out in the range of 50–1800  $\text{cm}^{-1}$  with an effective power of 6.0 mW, an accumulation number of 50, an exposure value of 7 s, and a resolution of 2  $\text{cm}^{-1}$ . Four spectral measurements were performed for the synthesized material and spectra have been processed (baseline correction and data smoothing) with the Spectra Manager™ software (Jasco).

ATR-IR spectra were recorded by the FTIR Vertex 70 interferometer (Bruker Optics, Ettlingen, Germany). The interferometer is equipped with an ATR crystal module (Bruker-Diamond single reflection crystal) with a single reflection diamond crystal at 45°, a thermal source (Glo-bar), and a wide-range RT-DLaTGS detector. The measurements were carried out in a vacuum. The ATR-IR spectra of  $\text{WO}_3$  powder were acquired with an acquisition rate of 7.5 kHz in the spectral region of 50–6000  $\text{cm}^{-1}$  by averaging 128 scans with a spectral resolution of 4  $\text{cm}^{-1}$ . Second-derivative analyses were performed both for the Raman and for the ATR-IR spectra by applying a 13-point smoothing filter [36]. Post-processing data analysis and visualization were performed using OPUS™ 8.2 software (Bruker) and OriginPro™ 9.0, respectively.

### 2.3. Fabrication of $\text{WO}_3$ sensor device

To study the sensing properties of the synthesized nanopowder, the alumina substrates (99.9% purity, 2 mm × 2 mm, Kyocera, Japan) were cleaned by sonication in acetone, ethanol, and  $\text{H}_2\text{O}$ . Afterward, TiW adhesion films and platinum interdigitated electrodes were deposited on the alumina substrates by means of DC magnetron sputtering at a power of 70 W, 7 SCCM argon flow, and  $5.1 \times 10^{-3}$  mbar pressure in the sputtering chamber. A platinum heater was deposited on the backside of the substrates to control the operating temperature of the sensing structure (Fig. S1). Then, the fabricated  $\text{WO}_3$  powder was dispersed in  $\text{H}_2\text{O}$  and sonicated for 10 min. The resulting dispersion was drop-casted on alumina substrates using a dispenser (Gilson Company, France). We made a cut of the substrate to show the thickness of the  $\text{WO}_3$  nanostructure after the drop-casting of the aqueous dispersion of nanoparticles (Fig. S2). As-prepared materials were annealed at 450 °C to stabilize their crystalline and electrical properties at high operating temperatures. Therefore, we chose 400 °C as the maximum temperature for gas sensing studies. Sensing analyses were carried out in a thermostatic test chamber, where the humidity level is kept under control using a humidity sensor. A detailed description of the system for gas sensing measurements has been reported previously [37].

The sensing response of the  $\text{WO}_3$  structure toward reducing gases was calculated according to Eqn (1):

$$\frac{(G_f - G_0)}{G_0} = \frac{\Delta G}{G_0} \quad (1)$$

Here,  $G_0$  is the baseline conductance of the  $WO_3$  sensor in air and  $G_f$  is the steady-state conductance value of the sensor in the presence of the analyte compound. The response toward an oxidizing gas was calculated according to Eqn (2), where  $R_0$  is the baseline resistance of the  $WO_3$  in air, and  $R_f$  is the steady-state resistance of the material in the presence of oxidizing gas.

$$\frac{(R_f - R_0)}{R_0} = \frac{\Delta R}{R_0} \quad (2)$$

### 3. Results and discussion

#### 3.1. Morphology and structure

Fig. 1 (a–d), (f), and (g) report the FESEM images of the  $WO_3$  materials prepared in  $H_2O$ , in the aqueous solutions of vitamin C and potassium sulfate. The morphology of the  $WO_3$  structure synthesized in the surfactant-free aqueous solution with a reaction time of 5 h is shown in Fig. 1 (a) and (b). Herein, agglomeration of particles is observed. Instead, well-defined  $WO_3$  nanoparticles with an average dimension of 52 nm are fabricated in the vitamin C-containing aqueous solution using the same reaction time (Fig. 1 (c–e) and Fig. S3 (a)). The morphology of the material prepared in the aqueous solution of potassium sulfate is shown in Fig. 1 (f) and (g). The obtained results indicate that the  $WO_3$  particles agglomerate.

The agglomeration of  $WO_3$  nanoparticles synthesized in  $H_2O$  (Fig. 1 (a) and (b)) can be attributed to the high reactivity of the tungsten precursor ( $WCl_6$ ) and its rapid hydrolyzation [38]. The morphological observations of the  $WO_3$  structure prepared in the aqueous solution of vitamin C indicate that the latest acts as a mild reducing and capping agent, resulting in uniform growth of  $WO_3$  particles without agglomeration [39,40]. In this case, the organic molecules attach to the growth surface and avoid the agglomeration of nanoparticles owing to the electrostatic or steric stabilization effects. Hence, vitamin C occupies the available growth sites having a significant effect on the formation and size of nanoparticles [41–45]. Furthermore, the application of potassium sulfate as a capping agent partially affects the separation of particles (Fig. 1 (f and g)), which indicates that vitamin C-containing aqueous solution is the most efficient for the synthesis of  $WO_3$  nanoparticles. In the aforementioned experiments, the reaction procedures for the fabrication of precipitates are performed at room temperature, thus reducing power consumption. We propose that the insufficient separation of particles in the aqueous solution of potassium sulfate can be described as follows: due to the low-temperature processing, the capping interaction is not strong enough, and therefore the primary  $WO_3$  nucleus can be formed without controlling the growth mechanism by the capping agent (potassium sulfate) [46]. However, it is worth mentioning that the nanoparticles tend to agglomerate when the reaction time in the solution of vitamin C is increased to 24 h (Fig. S2 (b) and (c)). This can be attributed to the self-assembly effect of the Ostwald ripening

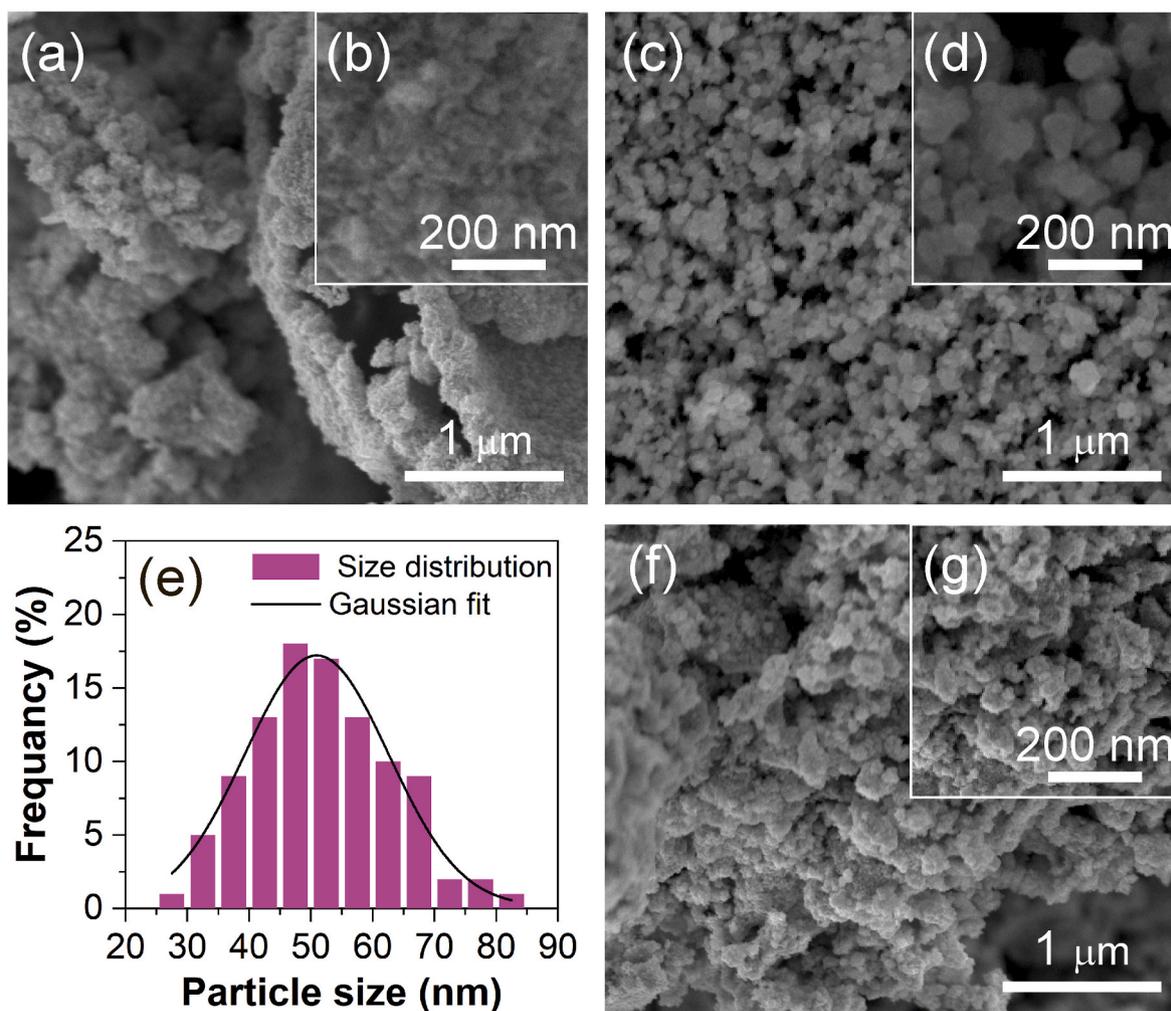


Fig. 1. FESEM images of fabricated  $WO_3$  powders: (a) and (b) Material synthesized in  $H_2O$  with a reaction time of 5 h, (c) and (d)  $WO_3$  nanoparticles prepared in vitamin C-containing aqueous solution with a reaction time of 5 h. (e) Particle size distribution histogram of  $WO_3$  powder prepared in the aqueous solution of vitamin C with a reaction time of 5 h. (f) and (g)  $WO_3$  material obtained in potassium sulfate-containing aqueous solutions with a reaction time of 5 h.

mechanism, where small particles dissolve in the solution followed by the formation of bigger ones [47,48]. Therefore, further, we studied the crystalline, compositional, and gas sensing properties of the  $\text{WO}_3$  nanopowder synthesized in the aqueous solutions of vitamin C with a reaction time of 5 h (Fig. 1 (c) and (d)). Moreover, vitamin C is a green agent, and thus the preparation of  $\text{WO}_3$  nanopowder in the aqueous solution of vitamin C can be considered a green synthesis procedure. Furthermore, previous works used vitamin C with other additives such as hydrochloric acid, urea, oxalic acid, and sodium sulfate, which are not considered environmentally friendly reagents [49–51].

Fig. 2 (a) shows the XRD pattern of the  $\text{WO}_3$  powder. The diffraction pattern of the material is matched with the monoclinic phase (JCPDS no. 98-007-1692, space group: P 121/c 1 and space group number: 14). No other phases are detected in this pattern. The main peaks located at 23.175, 23.653, 24.418, 28.843, 33.353, 34.277, 41.707, 50.088, 50.650, and 55.881 are attributed to the (100), (020), (002), (11–1), (120), (02–2), (122), (210), (12–3), and (14–1) planes of monoclinic  $\text{WO}_3$ , respectively.

The Raman spectrum of the  $\text{WO}_3$  powder is reported in Fig. 2 (b). The peak positions and their assignments are listed in Table 1 [52–55]. In particular, every mode can be associated with a crystalline phase or with a hydration state. We mainly compared our results with the detailed studies of Pecquenard et al. [53] and Daniel et al. [52]. Generally, the Raman spectrum of  $\text{WO}_3$  shows the contribution of deformation lattice mode between 200 and 400  $\text{cm}^{-1}$ , stretching modes between 600 and 900  $\text{cm}^{-1}$ , and then the contribution of external terminal oxygen bond  $\text{W}=\text{O}$  around 950  $\text{cm}^{-1}$  [52–55].

In the spectrum of prepared  $\text{WO}_3$  powder, intense Raman resonances are visible around 59, 71, 92, 134, 271, 329, 714, and 805  $\text{cm}^{-1}$ . The peak at 71  $\text{cm}^{-1}$  and the intense shoulders at 59 and 92  $\text{cm}^{-1}$  are due to  $(\text{W}_2\text{O}_2)_n$  chain vibrations in the lattice of  $\text{WO}_3$ . They are characteristic frequencies of the monoclinic structure [53,54]. The peak located around 270  $\text{cm}^{-1}$  can be assigned to O–W–O bending modes of binding oxygen of  $\text{WO}_3$  [55]. The weak peak at 188  $\text{cm}^{-1}$  is associated with the lattice mode and is characteristic of the monoclinic structure [52]. The sharp bands at 805  $\text{cm}^{-1}$  and 714  $\text{cm}^{-1}$  can be assigned to the stretching vibration of O–W–O and correspond to the mode of monoclinic  $\text{WO}_3$  [52]. The  $\text{WO}_3$  powder is synthesized using vitamin C ( $\text{C}_6\text{H}_8\text{O}_6$ ) as a surfactant and  $\text{H}_2\text{O}$  as a solvent. However, residual vitamin C content is not observed in its Raman spectrum.

Considering the Raman peak assignments, the presence of the bands at 59, 71, 92, 133, 188, 221, 270, 329, 436, 714, and 805  $\text{cm}^{-1}$  indicate that the prepared powder is mainly crystallized into the monoclinic  $\text{WO}_3$ . However, we observed very weak shoulders at high frequencies (between 940  $\text{cm}^{-1}$  and 960  $\text{cm}^{-1}$ ) and at low frequencies (at 207 and 375  $\text{cm}^{-1}$ ), which suggests the presence of a little amount of hydrates.

Fig. 2 (c) shows the ATR-IR spectrum of the fabricated  $\text{WO}_3$  powder. The peak positions and assignments of  $\text{WO}_3$  powder are reported in Table 2. In the far-IR region, the spectrum shows intense and visible broad bands at around 100 and 189  $\text{cm}^{-1}$ , which can be ascribed to the lattice modes of W-oxides. Furthermore, the main IR bands located in the very low-frequency region (around 160, 140, 130, and 110  $\text{cm}^{-1}$ ) can be ascribed to lattice modes of  $\text{WO}_3$ . Three bands at 242, 317 and around 338  $\text{cm}^{-1}$  correspond to the vibration modes of W–O–W [52, 57–59]. The IR bands at 318–360  $\text{cm}^{-1}$  correspond to the W–O–W vibration of  $\text{WO}_3$ . A shoulder is observed at 394  $\text{cm}^{-1}$ , which can be attributed to the  $(\text{W}-\text{OH}_2)$  bending modes of water molecules [57,59].

An intense broad band is observed in the mid-IR spectral region. Its main sharp peak at 616  $\text{cm}^{-1}$  and two shoulders around 730 and 757  $\text{cm}^{-1}$  can be associated with  $(\text{W}-\text{O})$  vibrational stretching modes [57, 59]. The other two shoulders, identified with the 2nd derivative analysis, are located at around 867 and 954  $\text{cm}^{-1}$  and indexed as stretching modes of  $(\text{O}-\text{W}-\text{O})$  and  $(\text{W}=\text{O})$ , respectively [58,60].

Furthermore, the ATR-IR spectrum of the  $\text{WO}_3$  powder showed a very weak band at around 1617  $\text{cm}^{-1}$  located in the bending region of  $\text{H}_2\text{O}$  molecules, which could be related to the water content in the

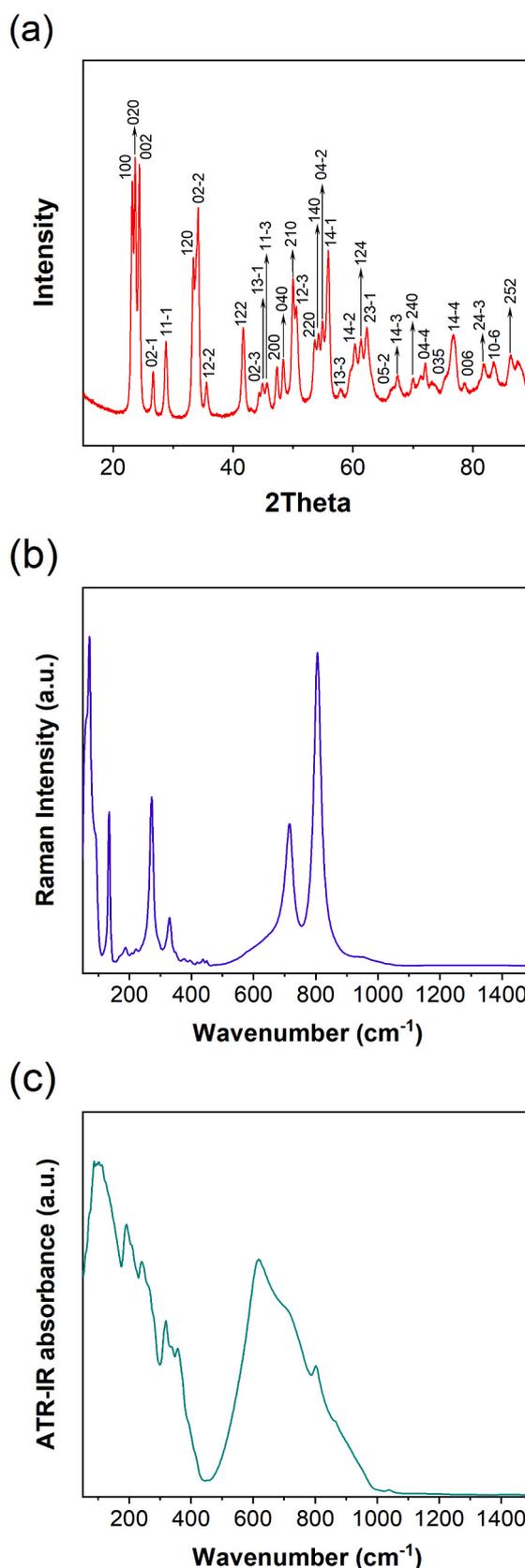


Fig. 2. (a) XRD pattern of the  $\text{WO}_3$  nanopowder synthesized in the aqueous solution of vitamin C with a reaction time of 5 h. (b) Raman spectrum of  $\text{WO}_3$  powder in the spectral region of 50–1500  $\text{cm}^{-1}$ . The whole Raman spectrum is reported in Fig. S4 (a). (c) ATR-IR spectrum of  $\text{WO}_3$  in the spectral region between 50 and 1500  $\text{cm}^{-1}$ . The whole ATR-IR spectrum is reported in Fig. S4 (b).

**Table 1**  
Raman peak/band positions and chemical group assignment of the fabricated WO<sub>3</sub> powder [52,56,57].

Frequency position cm <sup>-1</sup>	Assignment
59	$\nu$ (W <sub>2</sub> O <sub>2</sub> )
71	$\nu$ (W <sub>2</sub> O <sub>2</sub> )
92	$\nu$ (W <sub>2</sub> O <sub>2</sub> ) Lattice mode
133	Lattice mode
170	–
188	Lattice mode
207	$\nu$ (W–O–W)
221	$\nu$ (W–O–W)
270	$\delta$ (O–W–O)
329	$\delta$ (O–W–O) $\nu$ (W–OH <sub>2</sub> )
375	$\nu$ (W–OH <sub>2</sub> )
395	//weak shoulder
436	$\nu$ (W–OH <sub>2</sub> )
714	$\nu$ (O–W–O)
805	$\nu$ (O–W–O)
954	$\nu$ (W=O)

**Table 2**  
Infrared peak/band positions and assignments of WO<sub>3</sub> powder [52, 57–61].

Frequency position (cm <sup>-1</sup> )	Assignment
50	lattice modes
85	lattice modes
100	lattice modes
110	lattice modes
131	–
140	lattice modes
160	lattice modes
189	lattice modes
211	lattice modes
242	$\nu$ (W–O–W)
265	$\nu$ (W–O–W)
282	$\delta$ (O–W–O)
317	$\nu$ (W–O–W)
338	$\nu$ (W–O–W)
358	$\nu$ (W–OH <sub>2</sub> )
394	$\delta$ (W–OH <sub>2</sub> )
417	$\delta$ (W–O)
614	–
638	$\nu$ (W–O)
726	$\nu$ (W–O)
757	$\nu$ (O–W–O)
804	
867	$\nu$ (O–W–O)
954	$\nu$ (W=O)
1012	–
1039	$\delta$ (W–OH)
1617	$\delta$ (OH)

material [57].

The presence of the main bands at 282, 338, 638, 736, 757, and 867 cm<sup>-1</sup> can be associated with the monoclinic WO<sub>3</sub> phase [52,56,57], which is in agreement with the results obtained by the Raman measurements.

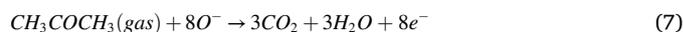
### 3.2. Gas sensing results

Fig. 3 (a) reports the sensing response of WO<sub>3</sub> nanoparticles toward 10 ppm of acetone at different operating temperatures (from 200 to 400 °C) and 40% of relative humidity (RH). As is apparent, the response of the material increases with its operating temperature showing the highest response value at 400 °C. Fig. 3 (b) displays the dynamic gas sensing characteristic of the WO<sub>3</sub> sensor at its optimum operating temperature (400 °C). The electrical conductance of the sensor increases due to the injection of each concentration of acetone into the test chamber and returns back to its baseline after the purification of the chamber by synthetic air flow. When the material is exposed to air, the

environmental oxygen is adsorbed on its surface trapping electrons from the conduction band and consequently forming oxygen ions (Eqn (3–6)) [62,63].



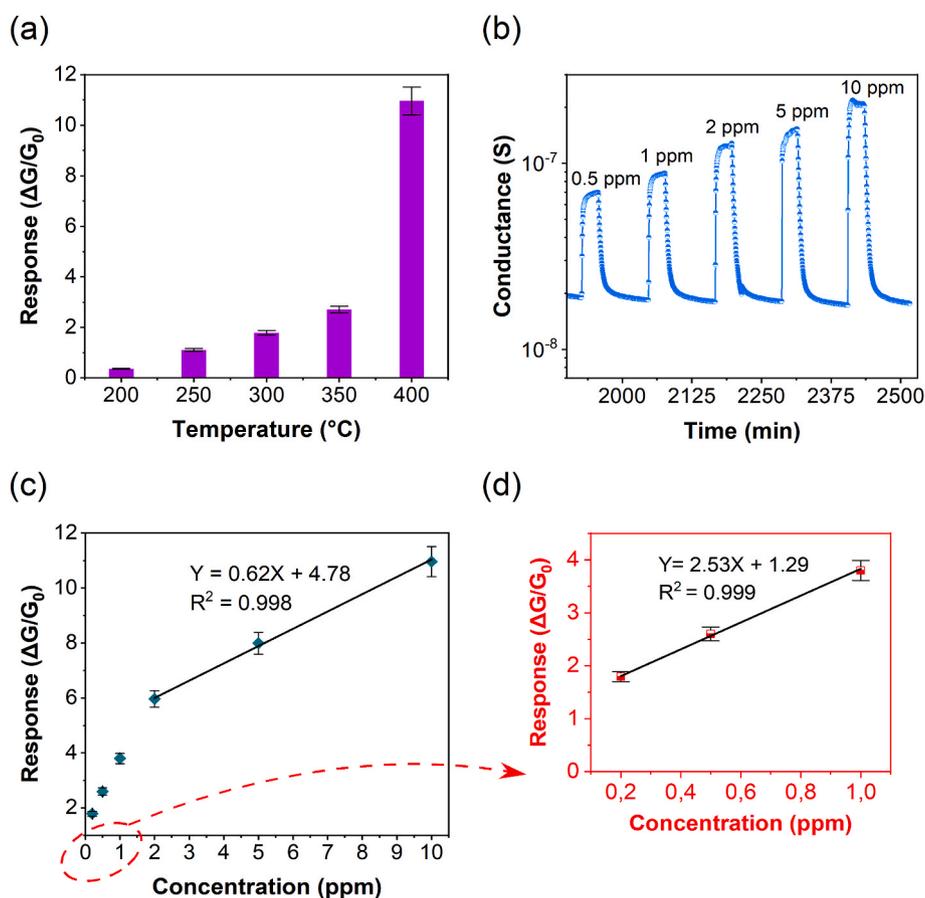
As the sensing structure is exposed to a reducing volatile organic compound such as acetone, the adsorbed oxygen species interact with its molecules (Eqn (7)), and the trapped electrons realize back to the conduction band of the material [64–66]. Hence, the adsorbed acetone acts as an electron donor and increases the conductance of the sensor.



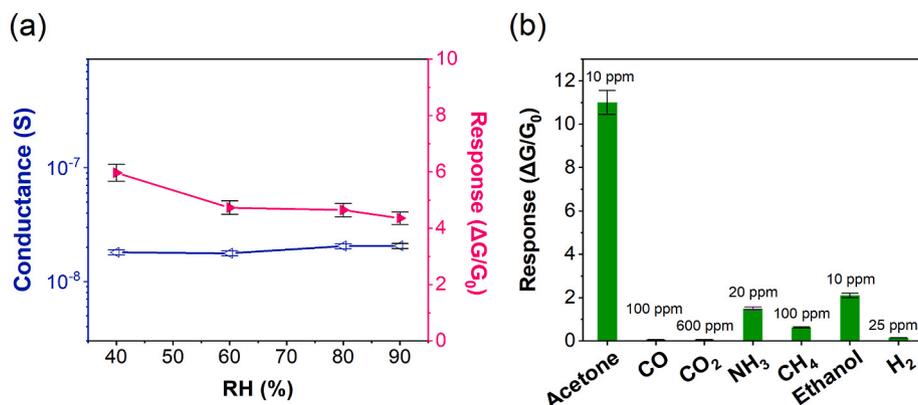
Furthermore, the conductance variation behavior of the sensor due to the exposure to acetone and purification of the test chamber indicates the reversible interaction between the acetone and fabricated nanostructure and confirms the n-type conductivity of WO<sub>3</sub> [67]. The adsorption of environmental oxygen on the surface of WO<sub>3</sub> is less efficient at low operating temperatures, which may affect the interaction and exchange process of charge carriers between acetone molecules and the material surface [28,68]. Furthermore, the chemisorption of complex organic molecules of acetone on oxide material requires high activation energy and multi-step reactions between the sensing structure and acetone [31,69]. Therefore, the response of the fabricated WO<sub>3</sub> sensor to acetone strongly depends on its operating temperature and shows an optimum value at 400 °C (Fig. 3(a)).

Response variation of the WO<sub>3</sub> depending on the concentration of acetone (0.2–10 ppm) is demonstrated in Fig. 3 (c) and (d) (The response values of the material are reported in Table S1). Here, two linear regions can be observed. Curve-fitting analyses using linear regression show that the slope (2.53) for the low concentrations range (0.2–1 ppm) is larger compared to the one (0.62) for the higher concentrations range (2–10). These findings indicate that the material exhibits a higher sensitivity at low concentrations of acetone. The good response of the material to low concentrations of acetone is a significant achievement because the concentration level of breath acetone for healthy people is lower than 1.8 ppm and for diabetics is higher compared to the aforementioned value [70,71]. Thus, the synthesized material is promising for its application in medical diagnostics and environmental monitoring.

Typically, the properties of chemiresistive sensing structures should be investigated at different levels of humidity, which is very important for their real-life applications. Therefore, we analyzed the functionalities of the material increasing the concentration of RH in the test chamber. The electrical conductance and the sensing response of the WO<sub>3</sub> nanostructure as a function of RH are shown in Fig. 4 (a). There is not a significant variation in the conductance of WO<sub>3</sub> with increasing the concentration of RH from 40 to 60%. The conductance value slightly increases (14%) at 80 and 90% of RH. Good stability of electrical conductance of the structure at different RH levels can be attributed to enhanced desorption of water molecules at 400 °C [62,63]. The response of the sensor is reduced by 17% due to the gradual increase of RH in the test chamber from 40 to 90%. Thus, the adsorption of water on WO<sub>3</sub> and hydroxylation of its surface partially affect the electrical conductance of the sensor and its interaction with the analyte molecules. However, the competitive adsorption of water and acetone on the WO<sub>3</sub> surface is a very complex process. In this case, the adsorbed water molecules may affect the formation and the number of oxidized species for the adsorption of acetone [63,64]. Meanwhile, the ionization of water molecules on the surface of the material could be an additional factor partially influencing the response of the sensor [72]. Hence, the small variation in the sensing response of WO<sub>3</sub> with the concentration of RH



**Fig. 3.** (a) Acetone sensing response of WO<sub>3</sub> nanomaterial at different temperatures. (b) Dynamic response of the WO<sub>3</sub> gas sensor toward acetone (0.5, 1, 2, 5, and 10 ppm) at 400 °C (RH, 40%). (c) Response of WO<sub>3</sub> toward different concentrations of acetone (0.2–10 ppm) at an operating temperature of 400 °C. (d) A magnified section from figure (c) where the response of the material toward low concentrations of acetone (from 200 ppb to 1 ppm) can be clearly seen.



**Fig. 4.** (a) The variation of the conductance and response values (toward 2 ppm of acetone) of WO<sub>3</sub> depending on the concentration of RH (40–90%) in the test chamber. (b) Respon of WO<sub>3</sub> to acetone, CO, CO<sub>2</sub>, NH<sub>3</sub>, CH<sub>4</sub>, ethanol, and H<sub>2</sub> at 400 °C.

can be attributed to the abovementioned processes.

The ability of a sensing material to determine a specific gaseous compound is an important feature. Therefore, we analyzed the selectivity of the WO<sub>3</sub> sensor to other compounds such as carbon monoxide (CO), carbon dioxide (CO<sub>2</sub>), ammonia (NH<sub>3</sub>), methane (CH<sub>4</sub>), ethanol, and hydrogen (H<sub>2</sub>). The obtained results are reported in Fig. 4 (b). The structure is almost not sensitive to CO, CO<sub>2</sub>, CH<sub>4</sub>, and H<sub>2</sub>. Furthermore, its response to 20 ppm of NH<sub>3</sub> (response value: 1.5) and 10 ppm of ethanol (response value: 2.1) is significantly lower compared to 10 ppm of acetone (response value: 11). Thus, the synthesized WO<sub>3</sub> material

shows an excellent selective response to acetone. The fabricated monoclinic WO<sub>3</sub> nanostructure (Fig. 2) has a strong interaction with polar molecules of acetone due to its surface acidity and large dipole moment [73–75]. This could be the reason for the high response of the sensor to acetone. Instead, the hexagonal WO<sub>3</sub> is more suitable for the adsorption of CO and CH<sub>4</sub> [76,77]. CO<sub>2</sub> is a linear molecule having a zero dipole moment [78]. The weak interaction of monoclinic WO<sub>3</sub> with CO<sub>2</sub> can be ascribed to the above-mentioned fact [79]. NH<sub>3</sub> and ethanol have smaller dipole moments [80,81]. It has also been demonstrated that the monoclinic WO<sub>3</sub> interacts better with acetone than with ethanol

owing to its surface acidity [82]. Furthermore, the excellent response of monoclinic WO<sub>3</sub> to acetone compared to ethanol, NH<sub>3</sub>, and H<sub>2</sub> can be due to differences in adsorption and desorption rates between the material and these gaseous compounds [83,84].

Table S2 reports a comparison between the acetone sensing parameters of the prepared WO<sub>3</sub> nanopowder and other semiconductor structures. Making a direct comparison of sensing structures is difficult, as different experimental setups and test protocols have been used. However, it is worth mentioning that the fabricated WO<sub>3</sub> nanostructure has a good and selective response to acetone without the use of dopants, noble metal catalytic layers, or other additives to functionalize its surface (Table S2). Thus, our findings and the comparison with other metal oxide-based acetone sensors indicate that the monoclinic WO<sub>3</sub> particles owing to their surface acidity and nanodimensions are suitable for the efficient detection of acetone molecules. Furthermore, the sensing tests after 2 months indicate that the WO<sub>3</sub> material has stable functional performance for acetone detection (Fig. S5). Consequently, the obtained results are important for the synthesis of WO<sub>3</sub> nanomaterials using green procedures and their application in acetone sensing devices.

#### 4. Conclusions

We synthesized WO<sub>3</sub> nanopowder by precipitation method using an environmentally friendly solution containing H<sub>2</sub>O as the solvent and vitamin C as a surfactant. The morphological analysis of the powder indicates that vitamin C acts as a mild reducing and capping agent in the aqueous solution, resulting in the formation of WO<sub>3</sub> nanoparticles without agglomeration. We also studied the effect of potassium sulfate on particle formation. However, the experimental findings suggest that vitamin C is more effective as a capping agent compared to potassium sulfate. Meanwhile, a further increase in the reaction time can lead to the agglomeration of particles in the aqueous solution of vitamin C. Thus, the reaction time is another significant parameter for the synthesis of WO<sub>3</sub> nanoparticles. Compositional and structural investigations of the prepared nanopowder indicate that it is monoclinic WO<sub>3</sub>, which plays a crucial role in its high and selective response to acetone. The enhancement in the sensor response as a function of its operating temperature can be attributed to the improvement of the adsorption of environmental oxygen on the WO<sub>3</sub> surface and required high activation energy for the chemisorption of acetone and its decomposition into CO<sub>2</sub> and H<sub>2</sub>O. The fabricated sensing structure is able to detect 200 ppb of acetone without doping or functionalization with specific materials and adding noble metal catalysts. It exhibits a considerably good and stable response to acetone even at 90% of RH. It is expected that the obtained results may be useful for the application of green procedures in the preparation of WO<sub>3</sub> nanomaterials and the development of acetone sensors, as well as other catalytic devices.

#### Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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#### Appendix B. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.ceramint.2023.06.314>.

#### Appendix A. Supplementary data

Supplementary data to this article can be found online at (...).

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