# Thickness-Dependent Dark-Bright Exciton Splitting and Phonon Bottleneck in CsPbBr<sub>3</sub>-Based Nanoplatelets Revealed via Magneto-Optical Spectroscopy

Shuli Wang, Mateusz Dyksik, Carola Lampe, Moritz Gramlich, Duncan K. Maude, Michał Baranowski, Alexander S. Urban,\* Paulina Plochocka,\* and Alessandro Surrente\*



process.

**KEYWORDS:** Metal halide perovskites, nanoplatelets, excitons, fine structure splitting, magneto-optical spectroscopy

he synthesis of colloidal nanocrystals with near-unity photoluminescence (PL) quantum yields has vastly extended the potential of metal halide perovskites for solidstate lighting and display applications.<sup>1-8</sup> This and their remarkably improved long-term stability<sup>9,10</sup> have allowed the fabrication of high external quantum efficiency LEDs, 11,12 lowthreshold lasers,<sup>13–15</sup> and single-photon emitters<sup>16–20</sup> with long exciton coherence times.<sup>19</sup> It is possible to template the growth of nanocrystals to form planar, ultrathin perovskite sheets embedded between long organic molecules, which stabilize the colloids, referred to as nanoplatelets.<sup>21-23</sup> The atomically thin quantum wells shown schematically in Figure 1a, together with the significant difference between the dielectric constants of the organic spacers and inorganic slabs, shift the exciton resonances toward the blue spectral region (essential for the realization of cost-efficient display devices<sup>24</sup>) and vastly increase the exciton binding energy, up to hundreds of meVs.<sup>25–27</sup>

Band-edge exciton states in metal-halide perovskites stem from the Coulomb interaction of a hole residing in the S-like valence band  $(J^{\rm h} = \frac{1}{2}, J_z^{\rm h} = \pm \frac{1}{2})$  and an electron residing in the split-off P-like conduction band  $(J^{\rm e} = \frac{1}{2}, J_z^{\rm e} = \pm \frac{1}{2})$ ,<sup>28</sup> where  $J^{\rm e,h}$  is the total angular momentum, and  $J_z^{\rm e,h}$  is its *z*component. The electron-hole exchange interaction leads to the fine structure splitting<sup>29,30</sup> with an optically inactive dark state with total angular momentum J = 0 and an optically active bright triplet with total angular momentum J = 1.<sup>31–34</sup> An accurate and reliable measurement of the energy spacing within the exciton manifold is not only critical from the fundamental physics viewpoint. The presence of a dark ground state<sup>35</sup> is potentially detrimental for applications of lead halide perovskites as light emitters. The exciton fine structure of perovskite nanocrystals has been recently the object of extensive investigation.<sup>16,17,19,35–43</sup> The substantially larger confinement of charge carriers in nanoplatelets greatly enhances the electron–hole exchange interaction<sup>44</sup> and is expected to increase the splitting between the bright and dark states significantly.<sup>45,46</sup> The fine structure of perovskite nanoplatelets has received considerably less attention, and experimental efforts have only recently focused on obtaining quantitative measurements.<sup>20,46–48</sup>

By its very nature, a dark state is inherently elusive to spectroscopic measurements. A commonly employed approach

Received:May 5, 2022Revised:August 20, 2022Published:August 29, 2022





**Figure 1.** (a) Schematic of the 2, 3, and 4 ML thick nanoplatelets together with the spatial dependence of the band gap and the dielectric constant. (b) STEM images of 3 ML nanoplatelets seen edge-on. (c-e) Absorption spectra of 2-4 ML thick nanoplatelets (continuous lines) fitted by 2D Elliott's formula (dashed gray lines). The contribution to the absorption spectrum from the continuum is shown as a shaded curve under each spectrum. (f) Energies of the 1s exciton transition ( $E_{1s}$ ) and continuum onset ( $E_c$ ). (g) Measured exciton binding energy ( $E_b$ ) as a function of nanoplatelet thickness. The red symbols (ML  $\geq$  5) are values taken from the literature.<sup>49</sup> The dashed line is drawn as a guide to the eye.

to evaluate the splitting between dark and bright excitons relies on the thermal mixing between these two populations and is determined through the temperature dependence of timeresolved PL.36,46,50-57 These results are often modeled with rate equations describing the recombination and relaxation processes within a three-level system, where the unknown quantities correspond to relaxation and recombination rates and the energy differences between the levels. However, this method is indirect,<sup>56</sup> and the three-level model might be oversimplified to describe certain systems accurately. Its results might indeed be affected by the presence of trap states, which introduce significantly smaller activation energies than the real dark bright splitting, 53,54 the interaction with surface dangling bonds,<sup>55</sup> or contributions from higher energy states.<sup>46,57</sup> In fact, instead of treating the dark-bright splitting as a free fitting parameter, in some cases it has been strongly constrained, or even fixed, relying on the outcome of alternative measurement methods. 46,51,56 In particular, in the case of perovskite nanoplatelets, due to the very large splittings expected within the exciton manifold, temperature-dependent time-resolved measurements alone have been considered unfit to provide a reliable quantitative estimation of the splittings, which were either fixed to values obtained by spectrally- and time-resolved measurements<sup>47</sup> or to values obtained by theoretical considerations.<sup>46</sup> This demonstrates the need for an experimental approach that yields straightforwardly the splitting between the dark and bright excitons without the need for background assumptions or complementary experimental methods.

An elegant way to overcome these limitations, providing direct spectroscopic access to dark exciton states, is to apply an in-plane magnetic field, which mixes the bright and dark states.<sup>35,58-61</sup> In a magnetic field, the dark state is expected to

display both a considerable increase of intensity and a redshift-two features that simultaneously occur only in this excitonic species. Thus, their observation allows us to unambiguously assign the corresponding peak to magnetic field brightened dark excitons without further modeling or assumptions, which are often required in the case of other approaches.<sup>36,37,52,62</sup>

In this Letter, we employ magnetic fields up to 65 T to investigate the exciton fine structure of  $Cs_{n-1}Pb_nBr_{3n+1}$ nanoplatelets with a thickness of n = 2-4 lead-halide octahedral planes (2-4 ML henceforth), shown schematically in Figure 1a. The magnetic-field-induced brightening enables us to determine the energy separation between dark and bright excitons accurately. Our measurements confirm that the dark state is consistently the lowest-lying exciton state in these nanoplatelets. The measured dark-bright splitting monotonically increases with decreasing thickness of the lead-halide octahedral plane, with values in the range  $\simeq 9-22$  meV. Additionally, our magneto-optical measurements allow us to shed light on the relaxation processes of excitons in lead halide perovskite nanoplatelets. In these colloidal nanocrystals, discrete, well-spaced states are expected to reduce the phonon scattering efficiency, which has been predicted to lead to a phonon bottleneck effect.<sup>63</sup> This effect has proven so far to be elusive in most experimental tests.<sup>64–67</sup> Recently, however, a dependence of the carrier relaxation rate on the nanocrystal size has been reported, which has been considered evidence for the phonon bottleneck in perovskite nanocrystals.<sup>68</sup> Our magneto-PL spectra, wherein the intensity of the dark and bright excitons suggests that the two populations are not fully thermalized to the lattice temperature, support the phonon bottleneck scenario.



**Figure 2.** Voigt configuration magneto-PL spectra of 3 ML nanoplatelets up to 65 T at 2 K. Spectra are shown at zero magnetic field (center) and in transversal (left) and longitudinal (right) polarization. Shading highlights the contribution of the PL peaks associated with the bright (blue) and dark excitons (gray). The spectra are normalized to the intensity of the zero magnetic field spectrum. (Inset) Schematic of the exciton manifold showing the selection rules.

In Figure 1b, we show an annular dark-field scanning transmission electron micrograph (STEM) of an ensemble of 3 ML thick nanoplatelets, demonstrating the high monodispersity of the synthesized sample. The zero-field absorption spectra measured at 5 K of 2, 3, and 4 ML thick nanoplatelets are displayed in Figure 1c-e. The samples were synthesized according to previously published methods.<sup>46,69</sup> Details on this and the experimental methods are given in the Supporting Information. The spectra display a single excitonic resonance, which we model with a generalized Elliott's formula to extract the 1s exciton transition energy and the onset of the continuum.<sup>49,70,71</sup> We plot the obtained 1s and continuum energies in Figure 1f as a function of the nanoplatelet thickness. A thinner inorganic slab increases the confinement of the charge carriers, which enhances the exciton binding energy, as shown in Figure 1g. Here, the exciton binding energy  $E_{\rm b} = E_{\rm c} - E_{\rm 1s}$  is the difference between the onset of the continuum absorption and the 1s exciton resonance. In weakly confined nanocrystals, with a side length of  $\simeq 10-12$  nm, the exciton binding energy<sup>49</sup> is comparable to that of the bulk material.7

We now turn to the fine structure of the exciton manifold in perovskite nanoplatelets, schematically presented in the inset of Figure 2. In nanoplatelets, the exchange interaction partially lifts the degeneracy of the bright exciton manifold.<sup>45,46</sup> Two bright states  $\phi_+$ ,  $\phi_-$  are degenerate and have in-plane dipole moments. These states, present in the case of cubic or tetragonal crystal structures,<sup>32,42,45</sup> couple to circularly polarized light. A lowering of the symmetry of the crystal structures lifts the degeneracy, giving rise to states that couple to two perpendicular linear polarization components. However, if it exists, this splitting is expected to be negligible compared to the dark-bright splitting,41,42,73 and we will neglect it in the following discussion. The third bright state,  $\phi_{Z'}$  has an out-of-plane dipole moment and couples to linearly polarized light, which propagates parallel to the nanoplatelet plane. The optically dark state is generally predicted to be the lowest-lying state. 31,32,35,42,46,58,61

In order to access the dark exciton state directly, we apply high in-plane magnetic field in the Voigt geometry. In this configuration, the zero-field exciton states are no longer eigenstates of the system. The new eigenstates can be obtained by considering linear combinations of the zero-field exciton states, with dipole moments oriented longitudinally (labeled L) or transversally (labeled T) to the magnetic field direction

$$\phi_{1,3L} = c_{1,3L}\phi_D + d_{1,3L}(\phi_+ - \phi_-)$$
(1a)

$$\phi_{2,3T} = c_{2,3T}\phi_Z + d_{2,3T}(\phi_+ + \phi_-)$$
(1b)

The coefficients are given by

$$c_{1,3L} = \frac{\frac{1}{\sqrt{2}}g_{L}\mu_{B}B}{\sqrt{2[E_{1,3L}(B) - E_{D}]^{2} + \frac{1}{2}(g_{L}\mu_{B}B)^{2}}}$$
(2a)

$$d_{1,3L} = \frac{E_{\rm D} - E_{1,3L}(B)}{\sqrt{2[E_{1,3L}(B) - E_{\rm D}]^2 + \frac{1}{2}(g_{\rm L}\mu_{\rm B}B)^2}}$$
(2b)

$$c_{2,3T} = \frac{\frac{1}{\sqrt{2}}g_{T}\mu_{B}B}{\sqrt{2[E_{2,3T}(B) - E_{Z}]^{2} + \frac{1}{2}(g_{T}\mu_{B}B)^{2}}}$$
(2c)

$$d_{2,3T} = \frac{E_{2,3T}(B) - E_Z}{\sqrt{2[E_{2,3T}(B) - E_Z]^2 + \frac{1}{2}(g_T \mu_B B)^2}}$$
(2d)

where  $g_L$  and  $g_T$  are the g-factors for the longitudinal and transverse modes.<sup>58,61,74</sup> The magnetic field also modifies the energy of the exciton resonances according to

$$E_{1,3L} = \frac{1}{2} \Big[ E_{\rm D} + E_{XY} \mp \sqrt{(E_{\rm D} - E_{XY})^2 + (g_{\rm L}\mu_{\rm B}B)^2} \Big]$$
(3a)  
$$E_{2,3T} = \frac{1}{2} \Big[ E_{Z} + E_{XY} \pm \sqrt{(E_{Z} - E_{XY})^2 + (g_{\rm T}\mu_{\rm B}B)^2} \Big]$$
(3b)

where  $E_D$ ,  $E_{XY}$ , and  $E_Z$  denote the energies of the dark state, the in-plane bright states, and the bright out-of-plane state, respectively.

According to eq 1a the magnetic field transfers the oscillator strength of the bright in-plane states to the dark exciton state. The magnetic field brightened dark state  $\phi_1$  couples to longitudinal polarized light (**B** || **E**) and its energy  $E_1$  decreases with increasing magnetic field, as predicted by eq 3a. The magnetic field also mixes the out-of-plane exciton state with



**Figure 3.** (a) Magneto-PL spectra in the Voigt configuration of 2 ML nanoplatelets, measured in longitudinal polarization ( $\mathbf{B} \parallel \mathbf{E}$ ) up to 64 T at 2 K. The peak attributed to the bright exciton (BX) is shown with blue shading, and the peak attributed to the dark exciton (DX) with gray shading. The vertical arrows are a guide to the eye. The transmission spectrum of the nanoplatelets is plotted in red. (b–d) Magnetic field dependence of the intensity of the dark and bright exciton transitions normalized to the intensity of the dark exciton transition at zero magnetic field for 2–4 ML thick nanoplatelets.

the in-plane bright states, see eq 1b. These states couple to transverse polarized light ( $\mathbf{B} \perp \mathbf{E}$ ) as indicated in the inset of Figure 2. The  $\phi_{3T}$  state is expected to red shift with increasing magnetic field, as shown by eq 3b.

Before presenting high magnetic field measurements, we focus on the zero-field PL spectrum of nanoplatelets. The PL spectrum of 3 ML thick nanoplatelets exhibits a maximum at  $\simeq 2.66$  eV and a shoulder on the low energy side (Figure 2, black curve). The spectrum of 2 ML thick nanoplatelets, shown in Figure 3a, consists of the main peak and a weaker high energy peak, which has an energy corresponding to the minimum of the transmission spectrum ( $\sim 2.90$  eV, see also Supporting Information). This strongly suggests that the high energy peak is related to the recombination of bright excitons (BX), the only species with a strong enough oscillator strength to yield a significant absorption at zero magnetic field. A similar PL line shape, with the simultaneous presence of two peaks attributed to dark and bright exciton states, was previously observed in 2 ML thick nanoplatelets.<sup>47</sup>

We move now to the description of our magneto-PL measurements, performed at  $\simeq 2$  K. We acquired magneto-PL spectra in the Voigt geometry with light which is linearly polarized transversally and longitudinally to the applied magnetic field. As seen in Figure 2 for 3 ML thick and Figure 3a for 2 ML thick nanoplatelets, in the longitudinal polarization, the lower energy PL component (gray shade) increases significantly in intensity with increasing magnetic field. We attribute this to the dark exciton (DX), which, as the lowest state, has a large thermal occupation, leading to a significant brightening when the oscillator strength is dramatically increased by the magnetic field-induced mixing of the states. This peak also exhibits a concomitant redshift, as predicted by eq 3a. These concurrent observations represent an unequivocal hallmark of the dipole-forbidden (dark) nature of the transition and provide very strong constraints in analyzing the magneto-PL spectrum. As shown in Figures 2 and 3a, the largest contribution to the PL spectrum in the high field limit stems from the brightened dark exciton peak. Beginning at high magnetic fields, we extract the energy of the dark exciton from the magneto-PL spectra as a function of the magnetic field only to values at which the peak energy can be determined with sufficiently high accuracy. This allows us to reliably fit the magnetic field dependence of the dark exciton energy using eq 3a. By extrapolating this dependence to zero

field, we can readily obtain the energy of the dark exciton at zero magnetic field without the need for analyzing in detail the corresponding zero-field PL spectrum. More details on the analysis of the magneto-PL spectra are provided in the Supporting Information.

Extrapolating the position of dark exciton peak to the low magnetic field limit, we note that it is also weakly present in the zero-field PL spectrum (Figure 2 and Figure 3a). This is consistent with prior reports on nanoplatelets<sup>46,47</sup> and Ruddlesden–Popper perovskites<sup>31,75–77</sup> and is probably related to a mixing of the dark and bright states due to spin–orbit interaction,<sup>31,76,78,79</sup> crystal distortions<sup>61,75,77</sup> or phonon-assisted transitions.<sup>61,75</sup> The higher energy PL band blue shifts and loses intensity with increasing in-plane magnetic field, as predicted by eq 3a. This confirms our assignment of this feature to emission from bright in-plane exciton states. More details on the fitting of the magneto-PL spectra, together with qualitatively similar magneto-PL of the 4 ML thick nanoplatelets, are provided in the Supporting Information.

It is important to note that our results suggest that the dark exciton is only partially responsible for the low energy PL spectrum at zero magnetic field. For example, for the 2 ML thick nanoplatelets, to fully describe the PL, we need to add additional transitions with maxima at ~2.86 and ~2.875 eV (dark blue and orange curves, see Figure 3a). The energy and emission intensity of these peaks are weakly dependent on the magnetic field strength. This suggests that these are related to the random potential landscape, due to the presence of unsaturated bonds at the boundary between the inorganic slabs and the ligands or at lattice defects, which can localize dark<sup>75</sup> excitons. These results suggest that the PL or bright' spectrum of CsPbBr<sub>3</sub>-based nanoplatelets is richer than initially thought, with features attributed to multiple bright, dark, and localized excitonic transitions. Similar considerations apply also to thicker nanoplatelets, as shown in the Supporting Information, where in Experimental Methods we also explain the origin of the differences between some of the spectral features reported here and those of pristine samples.

In the transversal polarization, we observe an overall reduction in the intensity of the high energy PL peak, accompanied by a slight red-shift, as shown in Figure 2 and in the analysis presented in the Supporting Information. This is expected for  $\phi_{3T}$  since the mixing of the bright in-plane states with the out-of-plane state (see eq 1b) reduces the oscillator



**Figure 4.** (a-c) Magnetic field dependence of the energies of the dark and bright exciton transitions relative to the zero-field transition energy of the bright state  $\Delta E$  for 2–4 ML nanoplatelets. The lines are fits of the data to eq 3a. (d) Measured bright-dark splitting as a function of nanoplatelet thickness. (e) PL intensity ratio between dark and bright exciton states for the three nanoplatelet thicknesses investigated as a function of the applied magnetic field. Full circles represent experimental points. The curves are calculated using eq 4, and the assumed temperature is indicated in the inset.

strength of the in-plane states. Additionally, the red-shift of this peak with increasing magnetic field is consistent with eq 3b (see Supporting Information for the complete magnetic field dependence of the PL energy on the magnetic field in transversal polarization). The large spectral broadening, together with the low thermal occupation, prevents us from observing the magnetic field brightened out-of-plane state ( $\phi_2$ ) at higher energies. The significantly lower thermal occupation of the out-of-plane state cannot be compensated by the transfer of oscillator strength from the in-plane states. This conclusion is also supported by theoretical considerations on the order of states in the bright exciton manifold.<sup>46</sup> The relaxed selection rules for the dark state allow for its observation also in the transversal polarization,<sup>75</sup> as shown in Figure 2 and discussed in the Supporting Information.

To corroborate our PL peak assignment, we analyze the intensities of the bright and dark exciton states (Figure 3b-d). We normalize all the peaks to the intensity of the dark exciton peak at zero magnetic field. The intensity of the dark state emission increases considerably, while that of the bright states decreases slightly with increasing magnetic field. This is fully consistent with the expected transfer of oscillator strength due to the magnetic field induced bright-dark mixing in the Voigt geometry. The 2 ML thick nanoplatelets exhibit the largest increase in PL intensity. This might appear surprising at first sight, considering that eq 2b predicts that a large dark-bright energy difference corresponds to a small gain in the dark excitons oscillator strength, as shown in the Supporting Information. However, the PL intensity is related to the product of the state occupation and the oscillator strength. Accordingly, a significantly larger population of the dark exciton state of the thinnest nanoplatelets can account for its larger magneto-PL intensity, as we discuss below in detail.

The magnetic field dependence of the energy of the bright and dark states, obtained by fitting the magneto-transmission spectra (see Supporting Information), is shown in Figure 4a-c. The predicted evolution using eq 3a reproduces well the observed blue and red-shifts and allows us to extract the darkbright splitting and a rough estimate of the longitudinal state's g-factor  $g_{\rm L} \simeq 1.2 - 2.3$  (see Supporting Information for details). The obtained dark-bright splitting, shown in Figure 4d, increases with decreasing thickness of the inorganic slab of nanoplatelets from ~9 meV for the 4 ML nanoplatelets to ~21 meV for the 2 ML nanoplatelets. The increase, as noted previously, is due to the larger dielectric and quantum confinement in the colloidal perovskite quantum wells.<sup>30,34,45</sup> The splittings reported here are slightly smaller than those measured in nanoplatelets synthesized with a very similar approach,<sup>46</sup> possibly due to a slightly larger in-plane size of the nanoplatelets investigated in our study and a concomitantly reduced in-plane confinement. However, the values are still considerably larger than the dark-bright splitting of conventional nanocubes,<sup>35,36</sup> which exhibit far less confinement.<sup>44</sup>

In Figure 4e, we plot the magnetic field dependence of the ratio between the intensity of the dark states,  $I_D$  and the bright states,  $I_B$ , which we calculate as

$$\frac{I_{\rm D}(B)}{I_{\rm B}(B)} = \frac{d_1^2(B)}{d_{\rm 3L}^2(B) + d_{\rm 3T}^2(B)} e^{-(E_1 - E_{\rm 3L})/k_B T}$$
(4)

where the Boltzmann factor accounts for the population ratio between the dark and bright states.<sup>61</sup> For simplicity, we neglect here the presence of localized exciton states. Due to the considerably more populated dark state in the 2 ML thick nanoplatelets, the PL intensity of the dark state is stronger in thinner nanoplatelets, as shown in Figure 4e. Importantly, to reproduce the experimental data, we had to use an effective exciton temperature of T = 15-47 K, as indicated in the inset of Figure 4e. These temperatures are at least an order of magnitude higher than the lattice temperature of  $\sim 2$  K. This mismatch between the lattice temperature and the effective exciton temperature suggests that the exciton population is not fully thermalized at the moment of the recombination.<sup>61</sup>

This observation can be explained in light of the energy level structure of the exciton manifold, in combination with the peculiar exciton-phonon coupling in metal halide perovskites. Due to the greatly enhanced exchange interaction in a strongly confined system,<sup>44</sup> the splittings in the exciton manifold exceed the energies of longitudinal optical (LO) phonons, with energies in the  $\sim 15-33$  meV range,  $^{69,80,81}$  to which excitons can couple to efficiently dissipate excess energy. In the case of the 4 ML thick nanoplatelets, a more efficient exciton relaxation to the dark state might be driven by an 8 meV LO phonon,<sup>80</sup> to which, however, carriers do not couple very efficiently.<sup>69</sup> This energy mismatch between LO phonons and dark-bright splittings, together with the virtually negligible coupling of carriers to acoustic phonons in metal halide perovskites,<sup>82</sup> leads to the highly nonthermalized exciton population visible in Figure 4e and supports the presence of a phonon bottleneck effect in strongly confined perovskite nanocrystals.68

In conclusion, we have used large in-plane magnetic fields to accurately extract the splitting between the dark and the bright exciton states of CsPbBr<sub>3</sub>-based nanoplatelets. In this configuration, the optically inactive states are brightened by the applied magnetic field and become the strongest contribution to the PL spectra while simultaneously red shifting with increasing magnetic field. The combination of these two observations allows us not only to accurately determine the splitting between the dark and bright excitons, even when this is considerably smaller than the inhomogeneous broadening of the PL spectrum, but also to reveal the presence of additional PL bands, possibly related to the localization of excitons. This approach, applied here for the first time to perovskite-based nanoplatelets, enabled us to confirm that the lowest-lying exciton state is optically dark and to determine the energy spacing between optically dark and optically bright states. The measured decrease in the darkbright splitting with increasing thickness of the nanoplatelets nicely reflects the expected increase in carrier confinement as the number of lead-halide octahedral planes is decreased. The energy splittings and magneto-PL intensities determined experimentally suggest that the dark and bright excitons are not fully thermalized with the crystal lattice. We attribute this observation to the mismatch between the large splittings within the exciton manifold and the energies of the LO phonons in metal halide perovskites. This, in combination with the poor coupling with acoustic phonons, leads to the observed nonthermalized exciton distribution, suggestive of the phonon bottleneck effect.

# ASSOCIATED CONTENT

#### **Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.nanolett.2c01826.

Experimental methods: nanoplatelets' synthesis and magneto-optical spectroscopy; note on the nanoplatelets' orientation; zero-field transmission and PL spectra; additional details concerning the analysis of the magneto-optical spectra in the longitudinal and transversal polarization (PDF)

## AUTHOR INFORMATION

#### **Corresponding Authors**

- Alexander S. Urban Nanospectroscopy Group and Center for Nanoscience (CeNS), Nano-Institute Munich, Department of Physics, Ludwig-Maximilians-Universität München (LMU), Munich 80539, Germany; <sup>©</sup> orcid.org/ 0000-0001-6168-2509; Email: urban@lmu.de
- Paulina Plochocka Laboratoire National des Champs Magnétiques Intenses, EMFL, CNRS UPR 3228, Université Grenoble Alpes, Université Toulouse, Université Toulouse 3, INSA-T, 38042 Grenoble and 31400 Toulouse, France; Department of Experimental Physics, Faculty of Fundamental Problems of Technology, Wroclaw University of Science and Technology, 50-370 Wroclaw, Poland; orcid.org/0000-0002-4019-6138; Email: paulina.plochocka@lncmi.cnrs.fr
- Alessandro Surrente Department of Experimental Physics, Faculty of Fundamental Problems of Technology, Wroclaw University of Science and Technology, 50-370 Wroclaw, Poland; orcid.org/0000-0003-4078-4965; Email: alessandro.surrente@pwr.edu.pl

#### Authors

- Shuli Wang Laboratoire National des Champs Magnétiques Intenses, EMFL, CNRS UPR 3228, Université Grenoble Alpes, Université Toulouse, Université Toulouse 3, INSA-T, 38042 Grenoble and 31400 Toulouse, France
- Mateusz Dyksik Department of Experimental Physics, Faculty of Fundamental Problems of Technology, Wroclaw University of Science and Technology, 50-370 Wroclaw, Poland; Occid.org/0000-0003-4945-8795
- Carola Lampe Nanospectroscopy Group and Center for Nanoscience (CeNS), Nano-Institute Munich, Department of Physics, Ludwig-Maximilians-Universität München (LMU), Munich 80539, Germany; orcid.org/0000-0001-7833-7306
- Moritz Gramlich Nanospectroscopy Group and Center for Nanoscience (CeNS), Nano-Institute Munich, Department of Physics, Ludwig-Maximilians-Universität München (LMU), Munich 80539, Germany; orcid.org/0000-0002-4733-4708
- Duncan K. Maude Laboratoire National des Champs Magnétiques Intenses, EMFL, CNRS UPR 3228, Université Grenoble Alpes, Université Toulouse, Université Toulouse 3, INSA-T, 38042 Grenoble and 31400 Toulouse, France
- Michał Baranowski Department of Experimental Physics, Faculty of Fundamental Problems of Technology, Wroclaw University of Science and Technology, 50-370 Wroclaw, Poland; © orcid.org/0000-0002-5974-0850

Complete contact information is available at: https://pubs.acs.org/10.1021/acs.nanolett.2c01826

#### Notes

The authors declare no competing financial interest.

# ACKNOWLEDGMENTS

This work has been partially funded by National Science Centre Poland within the SONATA program (Grant 2020/ 39/D/ST3/03000). The Polish participation in European Magnetic Field Laboratory (EMFL) is supported by the DIR/ WK/2018/07 grant from the Ministry of Science and Higher Education, Poland. This study has been partially supported through the EUR grant NanoX no. ANR-17-EURE-0009 in the framework of the "Programme des Investissements d'Avenir". S.W. holds a fellowship from the China Scholarship Council (CSC). M.D. acknowledges support from the Polish National Agency for Academic Exchange within the Bekker programme (Grant PPN/BEK/2019/1/00312/U/00001). A.S.U. acknowledges funding by the European Research Council Horizon 2020 through the ERC Grant Agreement PINNACLE (759744), by the Deutsche Forschungsgemeinsschaft (DFG) under Germanys Excellence Strategy EXC 2089/1-390776260 and by the Bavarian State Ministry of Science, Research and Arts through the grant "Solar Technologies go Hybrid (SolTech)".

# REFERENCES

(1) Tan, Z.-K.; Moghaddam, R. S.; Lai, M. L.; Docampo, P.; Higler, R.; Deschler, F.; Price, M.; Sadhanala, A.; Pazos, L. M.; Credgington, D.; Hanusch, F.; Bein, T.; Snaith, H. J.; Friend, R. H. Bright lightemitting diodes based on organometal halide perovskite. *Nat. Nanotechnol.* **2014**, *9*, 687–692.

(2) Lin, K.; et al. Perovskite light-emitting diodes with external quantum efficiency exceeding 20%. *Nature* **2018**, *562*, 245–248.

(3) Cao, Y.; et al. Perovskite light-emitting diodes based on spontaneously formed submicrometre-scale structures. *Nature* **2018**, 562, 249–253.

(4) Protesescu, L.; Yakunin, S.; Bodnarchuk, M. I.; Krieg, F.; Caputo, R.; Hendon, C. H.; Yang, R. X.; Walsh, A.; Kovalenko, M. V. Nanocrystals of cesium lead halide perovskites (CsPbX<sub>3</sub>, X = Cl, Br, and I): novel optoelectronic materials showing bright emission with wide color gamut. *Nano Lett.* **2015**, *15*, 3692–3696.

(5) Koscher, B. A.; Swabeck, J. K.; Bronstein, N. D.; Alivisatos, A. P. Essentially trapfree CsPbBr<sub>3</sub> colloidal nanocrystals by postsynthetic thiocyanate surface treatment. *J. Am. Chem. Soc.* **2017**, *139*, 6566–6569.

(6) Liu, F.; Zhang, Y.; Ding, C.; Kobayashi, S.; Izuishi, T.; Nakazawa, N.; Toyoda, T.; Ohta, T.; Hayase, S.; Minemoto, T.; Yoshino, K.; Dai, S.; Shen, Q. Highly luminescent phase-stable CsPbI<sub>3</sub> perovskite quantum dots achieving near 100% absolute photoluminescence quantum yield. *ACS Nano* **2017**, *11*, 10373–10383.

(7) Pan, J.; Shang, Y.; Yin, J.; De Bastiani, M.; Peng, W.; Dursun, I.; Sinatra, L.; El-Zohry, A. M.; Hedhili, M. N.; Emwas, A.-H.; Mohammed, O. F.; Ning, Z.; Bakr, O. M. Bidentate ligand-passivated CsPbI<sub>3</sub> perovskite nanocrystals for stable near-unity photoluminescence quantum yield and efficient red light-emitting diodes. J. Am. Chem. Soc. **2018**, 140, 562–565.

(8) Manoli, A.; Papagiorgis, P.; Sergides, M.; Bernasconi, C.; Athanasiou, M.; Pozov, S.; Choulis, S. A.; Bodnarchuk, M. I.; Kovalenko, M. V.; Othonos, A.; Itskos, G. Surface Functionalization of CsPbBr<sub>3</sub> Nanocrystals for Photonic Applications. *ACS Applied Nano Materials* **2021**, *4*, 5084–5097.

(9) Zheng, C.; Bi, C.; Huang, F.; Binks, D.; Tian, J. Stable and strong emission CsPbBr<sub>3</sub> quantum dots by surface engineering for high-performance optoelectronic films. *ACS Appl. Mater. Interfaces* **2019**, *11*, 25410–25416.

(10) Huang, H.; Zhao, W.; Yang, H.; Zhang, X.; Su, J.; Hu, K.; Nie, Z.; Li, Y.; Zhong, J. In situ synthesis of blue-emitting bromide-based perovskite nanoplatelets towards unity quantum efficiency and ultrahigh stability. *Journal of Materials Chemistry C* **2021**, *9*, 5535–5543.

(11) Dong, Y.; et al. Bipolar-shell resurfacing for blue LEDs based on strongly confined perovskite quantum dots. *Nat. Nanotechnol.* **2020**, *15*, 668–674.

(12) Fang, T.; Wang, T.; Li, X.; Dong, Y.; Bai, S.; Song, J. Perovskite QLED with an external quantum efficiency of over 21% by modulating electronic transport. *Science Bulletin* **2021**, *66*, 36–43.

(13) Huang, C.-Y.; Zou, C.; Mao, C.; Corp, K. L.; Yao, Y.-C.; Lee, Y.-J.; Schlenker, C. W.; Jen, A. K.; Lin, L. Y. CsPbBr<sub>3</sub> perovskite quantum dot vertical cavity lasers with low threshold and high stability. *ACS Photonics* **2017**, *4*, 2281–2289.

(14) Wang, Y.; Li, X.; Nalla, V.; Zeng, H.; Sun, H. Solutionprocessed low threshold vertical cavity surface emitting lasers from allinorganic perovskite nanocrystals. *Adv. Funct. Mater.* **2017**, *27*, 1605088.

(15) Shang, Q.; Li, M.; Zhao, L.; Chen, D.; Zhang, S.; Chen, S.; Gao, P.; Shen, C.; Xing, J.; Xing, G.; Shen, B.; Liu, X.; Zhang, Q. Role of the exciton-polariton in a continuouswave optically pumped CsPbBr<sub>3</sub> perovskite laser. *Nano Lett.* **2020**, *20*, 6636–6643.

(16) Park, Y.-S.; Guo, S.; Makarov, N. S.; Klimov, V. I. Room temperature single-photon emission from individual perovskite quantum dots. *ACS Nano* **2015**, *9*, 10386–10393.

(17) Rainò, G.; Nedelcu, G.; Protesescu, L.; Bodnarchuk, M. I.; Kovalenko, M. V.; Mahrt, R. F.; Stöferle, T. Single cesium lead halide perovskite nanocrystals at low temperature: fast single-photon emission, reduced blinking, and exciton fine structure. *ACS Nano* **2016**, *10*, 2485–2490.

(18) Fu, M.; Tamarat, P.; Trebbia, J.-B.; Bodnarchuk, M. I.; Kovalenko, M. V.; Even, J.; Lounis, B. Unraveling exciton-phonon coupling in individual FAPbI<sub>3</sub> nanocrystals emitting near-infrared single photons. *Nat. Commun.* **2018**, *9*, 3318.

(19) Utzat, H.; Sun, W.; Kaplan, A. E.; Krieg, F.; Ginterseder, M.; Spokoyny, B.; Klein, N. D.; Shulenberger, K. E.; Perkinson, C. F.; Kovalenko, M. V.; Bawendi, M. G. Coherent single-photon emission from colloidal lead halide perovskite quantum dots. *Science* **2019**, *363*, 1068–1072.

(20) Huo, C.; Fong, C. F.; Amara, M.-R.; Huang, Y.; Chen, B.; Zhang, H.; Guo, L.; Li, H.; Huang, W.; Diederichs, C.; Xiong, Q. Optical spectroscopy of single colloidal CsPbBr<sub>3</sub> perovskite nanoplatelets. *Nano Lett.* **2020**, *20*, 3673–3680.

(21) Sichert, J. A.; Tong, Y.; Mutz, N.; Vollmer, M.; Fischer, S.; Milowska, K. Z.; García Cortadella, R.; Nickel, B.; Cardenas-Daw, C.; Stolarczyk, J. K.; Urban, A. S.; Feldmann, J. Quantum size effect in organometal halide perovskite nanoplatelets. *Nano Lett.* **2015**, *15*, 6521–6527.

(22) Bekenstein, Y.; Koscher, B. A.; Eaton, S. W.; Yang, P.; Alivisatos, A. P. Highly luminescent colloidal nanoplates of perovskite cesium lead halide and their oriented assemblies. *J. Am. Chem. Soc.* **2015**, *137*, 16008–16011.

(23) Hintermayr, V. A.; Richter, A. F.; Ehrat, F.; Doblinger, M.; Vanderlinden, W.; Sichert, J. A.; Tong, Y.; Polavarapu, L.; Feldmann, J.; Urban, A. S. Tuning the Optical Properties of Perovskite Nanoplatelets through Composition and Thickness by Ligand-Assisted Exfoliation. *Adv. Mater.* **2016**, *28*, 9478–9485.

(24) Kim, S.; Kim, J.-A.; Kim, T.; Chung, H.; Park, S.; Choi, S.-M.; Kim, H.-M.; Chung, D.-Y.; Jang, E. Efficient blue-light-emitting Cd-free colloidal quantum well and its application in electroluminescent devices. *Chem. Mater.* **2020**, *32*, 5200–5207.

(25) Tanaka, K.; Takahashi, T.; Kondo, T.; Umebayashi, T.; Asai, K.; Ema, K. Image charge effect on two-dimensional excitons in an inorganic-organic quantum-well crystal. *Phys. Rev. B* 2005, *71*, 045312.

(26) Ji, B.; Rabani, E.; Efros, A. L.; Vaxenburg, R.; Ashkenazi, O.; Azulay, D.; Banin, U.; Millo, O. Dielectric confinement and excitonic effects in two-dimensional nanoplatelets. *ACS Nano* **2020**, *14*, 8257– 8265.

(27) Shornikova, E. V.; Yakovlev, D. R.; Gippius, N. A.; Qiang, G.; Dubertret, B.; Khan, A. H.; Di Giacomo, A.; Moreels, I.; Bayer, M. Exciton Binding Energy in CdSe Nanoplatelets Measured by One-and Two-Photon Absorption. *Nano Lett.* **2021**, *21*, 10525–10531.

(28) Tanaka, K.; Takahashi, T.; Ban, T.; Kondo, T.; Uchida, K.; Miura, N. Comparative study on the excitons in lead-halide-based perovskite-type crystals CH<sub>3</sub>NH<sub>3</sub>PbBr<sub>3</sub> CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub>. *Solid State Commun.* **2003**, *127*, 619–623.

(29) Efros, A. L.; Rosen, M.; Kuno, M.; Nirmal, M.; Norris, D. J.; Bawendi, M. Band-edge exciton in quantum dots of semiconductors with a degenerate valence band: Dark and bright exciton states. *Phys. Rev. B* **1996**, *54*, 4843.

(30) Fu, H.; Wang, L.-W.; Zunger, A. Excitonic exchange splitting in bulk semiconductors. *Phys. Rev. B* **1999**, *59*, 5568.

(31) Tanaka, K.; Takahashi, T.; Kondo, T.; Umeda, K.; Ema, K.; Umebayashi, T.; Asai, K.; Uchida, K.; Miura, N. Electronic and excitonic structures of inorganic–organic perovskite-type quantumwell crystal  $(C_4H_9NH_3)_2PbBr_4$ . *Jpn. J. Appl. Phys.* **2005**, *44*, 5923.

(32) Yu, Z. Effective-mass model and magneto-optical properties in hybrid perovskites. *Sci. Rep.* **2016**, *6*, 28576.

(33) Sercel, P. C.; Lyons, J. L.; Wickramaratne, D.; Vaxenburg, R.; Bernstein, N.; Efros, A. L. Exciton fine structure in perovskite nanocrystals. *Nano Lett.* **2019**, *19*, 4068–4077.

(34) Ben Aich, R.; Ben Radhia, S.; Boujdaria, K.; Chamarro, M.; Testelin, C. Multiband k·p model for tetragonal crystals: application to hybrid halide perovskite nanocrystals. *J. Phys. Chem. Lett.* **2020**, *11*, 808–817.

(35) Tamarat, P.; Bodnarchuk, M. I.; Trebbia, J.-B.; Erni, R.; Kovalenko, M. V.; Even, J.; Lounis, B. The ground exciton state of formamidinium lead bromide perovskite nanocrystals is a singlet dark state. *Nat. Mater.* **2019**, *18*, 717–724.

(36) Chen, L.; Li, B.; Zhang, C.; Huang, X.; Wang, X.; Xiao, M. Composition-dependent energy splitting between bright and dark excitons in lead halide perovskite nanocrystals. *Nano Lett.* **2018**, *18*, 2074–2080.

(37) Xu, K.; Vliem, J. F.; Meijerink, A. Long-lived dark exciton emission in Mn-doped CsPbCl<sub>3</sub> perovskite nanocrystals. *J. Phys. Chem. C* 2019, 123, 979–984.

(38) Hu, F.; Zhang, H.; Sun, C.; Yin, C.; Lv, B.; Zhang, C.; Yu, W. W.; Wang, X.; Zhang, Y.; Xiao, M. Superior optical properties of perovskite nanocrystals as single photon emitters. *ACS Nano* **2015**, *9*, 12410–12416.

(39) Yin, C.; Chen, L.; Song, N.; Lv, Y.; Hu, F.; Sun, C.; William, W. Y.; Zhang, C.; Wang, X.; Zhang, Y.; Xiao, M. Bright-exciton fine-structure splittings in single perovskite nanocrystals. *Phys. Rev. Lett.* **2017**, *119*, 026401.

(40) Becker, M. A.; et al. Bright triplet excitons in caesium lead halide perovskites. *Nature* **2018**, *553*, 189–193.

(41) Ramade, J.; Andriambariarijaona, L. M.; Steinmetz, V.; Goubet, N.; Legrand, L.; Barisien, T.; Bernardot, F.; Testelin, C.; Lhuillier, E.; Bramati, A.; Chamarro, M. Fine structure of excitons and electron-hole exchange energy in polymorphic CsPbBr<sub>3</sub> single nanocrystals. *Nanoscale* **2018**, *10*, 6393–6401.

(42) Fu, M.; Tamarat, P.; Huang, H.; Even, J.; Rogach, A. L.; Lounis, B. Neutral and charged exciton fine structure in single lead halide perovskite nanocrystals revealed by magneto-optical spectroscopy. *Nano Lett.* **2017**, *17*, 2895–2901.

(43) Qiao, T.; Liu, X.; Rossi, D.; Khurana, M.; Lin, Y.; Wen, J.; Cheon, J.; Akimov, A. V.; Son, D. H. Magnetic Effect of Dopants on Bright and Dark Excitons in Strongly Confined Mn-Doped CsPbI<sub>3</sub> Quantum Dots. *Nano Lett.* **2021**, *21*, 9543–9550.

(44) Takagahara, T. Effects of dielectric confinement and electronhole exchange interaction on excitonic states in semiconductor quantum dots. *Phys. Rev. B* **1993**, *47*, 4569.

(45) Ghribi, A.; Aich, R. B.; Boujdaria, K.; Barisien, T.; Legrand, L.; Chamarro, M.; Testelin, C. Dielectric Confinement and Exciton Fine Structure in Lead Halide Perovskite Nanoplatelets. *Nanomaterials* **2021**, *11*, 3054.

(46) Gramlich, M.; Swift, M. W.; Lampe, C.; Lyons, J. L.; Döblinger, M.; Efros, A. L.; Sercel, P. C.; Urban, A. S. Dark and Bright Excitons in Halide Perovskite Nanoplatelets. *Advanced Science* **2022**, *9*, 2103013.

(47) Rossi, D.; Liu, X.; Lee, Y.; Khurana, M.; Puthenpurayil, J.; Kim, K.; Akimov, A. V.; Cheon, J.; Son, D. H. Intense Dark Exciton Emission from Strongly Quantum-Confined CsPbBr<sub>3</sub> Nanocrystals. *Nano Lett.* **2020**, *20*, 7321–7326.

(48) Schmitz, A.; Montanarella, F.; Schaberg, L. L.; Abdelbaky, M.; Kovalenko, M. V.; Bacher, G. Optical Probing of Crystal Lattice Configurations in Single CsPbBr<sub>3</sub> Nanoplatelets. *Nano Lett.* **2021**, *21*, 9085–9092.

(49) Bohn, B. J.; Tong, Y.; Gramlich, M.; Lai, M. L.; Döblinger, M.; Wang, K.; Hoye, R. L.; Müller-Buschbaum, P.; Stranks, S. D.; Urban, A. S.; Polavarapu, L.; Feldmann, J. Boosting tunable blue luminescence of halide perovskite nanoplatelets through postsynthetic surface trap repair. *Nano Lett.* **2018**, *18*, 5231–5238.

(50) Labeau, O.; Tamarat, P.; Lounis, B. Temperature dependence of the luminescence lifetime of single CdSe/ZnS quantum dots. *Phys. Rev. Lett.* **2003**, *90*, 257404.

(51) Biadala, L.; Louyer, Y.; Tamarat, P.; Lounis, B. Direct observation of the two lowest exciton zero-phonon lines in single CdSe/ZnS nanocrystals. *Phys. Rev. Lett.* **2009**, *103*, 037404.

(52) Biadala, L.; Liu, F.; Tessier, M. D.; Yakovlev, D. R.; Dubertret, B.; Bayer, M. Recombination dynamics of band edge excitons in quasi-two-dimensional CdSe nanoplatelets. *Nano Lett.* **2014**, *14*, 1134–1139.

(53) de Mello Donegá, C.; Bode, M.; Meijerink, A. Size-and temperature-dependence of exciton lifetimes in CdSe quantum dots. *Phys. Rev. B* **2006**, *74*, 085320.

(54) Crooker, S.; Barrick, T.; Hollingsworth, J.; Klimov, V. Multiple temperature regimes of radiative decay in CdSe nanocrystal quantum dots: Intrinsic limits to the dark-exciton lifetime. *Appl. Phys. Lett.* **2003**, *82*, 2793–2795.

(55) Biadala, L.; Shornikova, E. V.; Rodina, A. V.; Yakovlev, D. R.; Siebers, B.; Aubert, T.; Nasilowski, M.; Hens, Z.; Dubertret, B.; Efros, A. L.; Bayer, M. Magnetic polaron on dangling-bond spins in CdSe colloidal nanocrystals. *Nat. Nanotechnol.* **2017**, *12*, 569–574.

(56) Shornikova, E. V.; et al. Addressing the exciton fine structure in colloidal nanocrystals: the case of CdSe nanoplatelets. *Nanoscale* **2018**, *10*, 646–656.

(57) Biadala, L.; Siebers, B.; Beyazit, Y.; Tessier, M. D.; Dupont, D.; Hens, Z.; Yakovlev, D. R.; Bayer, M. Band-edge exciton fine structure and recombination dynamics in InP/ZnS colloidal nanocrystals. *ACS Nano* **2016**, *10*, 3356–3364.

(58) Kataoka, T.; Kondo, T.; Ito, R.; Sasaki, S.; Uchida, K.; Miura, N. Magneto-optical study on excitonic spectra in  $(C_6H_{13}NH_3)_2PbI_4$ . *Phys. Rev. B* **1993**, 47, 2010.

(59) Bayer, M.; Ortner, G.; Stern, O.; Kuther, A.; Gorbunov, A.; Forchel, A.; Hawrylak, P.; Fafard, S.; Hinzer, K.; Reinecke, T.; Walck, S.; Reithmaier, J.; Klopf, F.; Scäfer, F. Fine structure of neutral and charged excitons in self-assembled In(Ga)As/(Al)GaAs quantum dots. *Phys. Rev. B* **2002**, *65*, 195315.

(60) Robert, C.; Han, B.; Kapuscinski, P.; Delhomme, A.; Faugeras, C.; Amand, T.; Molas, M. R.; Bartos, M.; Watanabe, K.; Taniguchi, T.; Urbaszek, B.; Potemski, M.; Marie, X. Measurement of the spinforbidden dark excitons in MoS<sub>2</sub> and MoSe<sub>2</sub> monolayers. *Nat. Commun.* **2020**, *11*, 4037.

(61) Dyksik, M.; Duim, H.; Maude, D. K.; Baranowski, M.; Loi, M. A.; Plochocka, P. Brightening of dark excitons in 2D perovskites. *Science Advances* **2021**, *7*, eabk0904.

(62) Canneson, D.; Shornikova, E. V.; Yakovlev, D. R.; Rogge, T.; Mitioglu, A. A.; Ballottin, M. V.; Christianen, P. C.; Lhuillier, E.; Bayer, M.; Biadala, L. Negatively charged and dark excitons in CsPbBr<sub>3</sub> perovskite nanocrystals revealed by high magnetic fields. *Nano Lett.* **2017**, *17*, 6177–6183.

(63) Forde, A.; Inerbaev, T.; Hobbie, E. K.; Kilin, D. S. Excited-state dynamics of a CsPbBr<sub>3</sub> nanocrystal terminated with binary ligands: sparse density of states with giant spin-orbit coupling suppresses carrier cooling. *J. Am. Chem. Soc.* **2019**, *141*, 4388–4397.

(64) Makarov, N. S.; Guo, S.; Isaienko, O.; Liu, W.; Robel, I.; Klimov, V. I. Spectral and dynamical properties of single excitons, biexcitons, and trions in cesium–lead-halide perovskite quantum dots. *Nano Lett.* **2016**, *16*, 2349–2362.

(65) Chen, J.; Messing, M. E.; Zheng, K.; Pullerits, T. Cationdependent hot carrier cooling in halide perovskite nanocrystals. *J. Am. Chem. Soc.* **2019**, *141*, 3532–3540.

(66) Mondal, A.; Aneesh, J.; Ravi, V. K.; Sharma, R.; Mir, W. J.; Beard, M. C.; Nag, A.; Adarsh, K. Ultrafast exciton many-body interactions and hot-phonon bottleneck in colloidal cesium lead halide perovskite nanocrystals. *Phys. Rev. B* **2018**, *98*, 115418.

(67) Li, Y.; Lai, R.; Luo, X.; Liu, X.; Ding, T.; Lu, X.; Wu, K. On the absence of a phonon bottleneck in strongly confined CsPbBr<sub>3</sub> perovskite nanocrystals. *Chemical science* **2019**, *10*, 5983–5989.

(68) Dai, L.; Deng, Z.; Auras, F.; Goodwin, H.; Zhang, Z.; Walmsley, J. C.; Bristowe, P. D.; Deschler, F.; Greenham, N. C. Slow carrier relaxation in tin-based perovskite nanocrystals. *Nat. Photonics* **2021**, *15*, 696–702.

(69) Gramlich, M.; Lampe, C.; Drewniok, J.; Urban, A. S. How Exciton–Phonon Coupling Impacts Photoluminescence in Halide Perovskite Nanoplatelets. *J. Phys. Chem. Lett.* **2021**, *12*, 11371–11377.

(70) Campi, D.; Coriasso, C. Optical nonlinearities in multiple quantum wells: Generalized Elliott formula. *Phys. Rev. B* 1995, *51*, 10719.

(71) Chen, X.; Lu, H.; Wang, K.; Zhai, Y.; Lunin, V.; Sercel, P. C.; Beard, M. C. Tuning Spin-Polarized Lifetime in Two-Dimensional Metal-Halide Perovskite through Exciton Binding Energy. *J. Am. Chem. Soc.* **2021**, *143*, 19438–19445.

(72) Yang, Z.; Surrente, A.; Galkowski, K.; Miyata, A.; Portugall, O.; Sutton, R.; Haghighirad, A.; Snaith, H.; Maude, D.; Plochocka, P.; Nicholas, R. J. Impact of the halide cage on the electronic properties of fully inorganic cesium lead halide perovskites. *ACS Energy Letters* **2017**, *2*, 1621–1627.

(73) Do, T. T. H.; Granados del Águila, A.; Zhang, D.; Xing, J.; Liu, S.; Prosnikov, M.; Gao, W.; Chang, K.; Christianen, P. C.; Xiong, Q. Bright exciton fine-structure in two-dimensional lead halide perovskites. *Nano Lett.* **2020**, *20*, 5141–5148.

(74) Surrente, A.; Baranowski, M.; Plochocka, P. Perspective on the physics of twodimensional perovskites in high magnetic field. *Appl. Phys. Lett.* **2021**, *118*, 170501.

(75) Ema, K.; Umeda, K.; Toda, M.; Yajima, C.; Arai, Y.; Kunugita, H.; Wolverson, D.; Davies, J. Huge exchange energy and fine structure of excitons in an organic-inorganic quantum well material. *Phys. Rev. B* **2006**, *73*, 241310.

(76) Wang, Y.; Song, F.; Yuan, Y.; Dang, J.; Xie, X.; Sun, S.; Yan, S.; Hou, Y.; Lou, Z.; Xu, X. Strong Triplet-Exciton–LO-Phonon Coupling in Two-Dimensional Layered Organic–Inorganic Hybrid Perovskite Single Crystal Microakes. J. Phys. Chem. Lett. **2021**, 12, 2133–2141.

(77) Choi, W.; Nam, S. H.; So, H.-K.; Lee, S.-E.; Jung, M.-H.; Jang, J. I. Impact of Dark Excitons on the Population and Relaxation Kinetics of Two-Dimensional Biexcitons in  $[CH_3(CH_2)_3NH_3]_2Pb_{1-x}Mn_xBr_4$  (x = 0 - 0.09). J. Am. Chem. Soc. **2021**, 143, 19785–19793.

(78) Goto, T.; Taguchi, S.; Cho, K.; Nagamune, Y.; Takeyama, S.; Miura, N. Magnetooptical effect of the Wannier exciton in a biaxial ZnP<sub>2</sub> crystal. III. *J. Phys. Soc. Jpn.* **1990**, *59*, 773–778.

(79) Goto, T.; Makino, H.; Yao, T.; Chia, C.; Makino, T.; Segawa, Y.; Mousdis, G. A.; Papavassiliou, G. C. Localization of triplet excitons and biexcitons in the two-dimensional semiconductor  $(CH_3C_6H_4CH_2NH_3)_2PbBr_4$ . *Phys. Rev. B* **2006**, 73, 115206.

(80) Du, W.; Zhang, S.; Wu, Z.; Shang, Q.; Mi, Y.; Chen, J.; Qin, C.; Qiu, X.; Zhang, Q.; Liu, X. Unveiling lasing mechanism in CsPbBr<sub>3</sub> microsphere cavities. *Nanoscale* **2019**, *11*, 3145–3153.

(81) Straus, D. B.; Hurtado Parra, S.; Iotov, N.; Gebhardt, J.; Rappe, A. M.; Subotnik, J. E.; Kikkawa, J. M.; Kagan, C. R. Direct observation of electron-phonon coupling and slow vibrational relaxation in organic-inorganic hybrid perovskites. *J. Am. Chem. Soc.* **2016**, *138*, 13798-13801.

(82) Straus, D. B.; Iotov, N.; Gau, M. R.; Zhao, Q.; Carroll, P. J.; Kagan, C. R. Longer cations increase energetic disorder in excitonic 2D hybrid perovskites. *J. Phys. Chem. Lett.* **2019**, *10*, 1198–1205.

# Recommended by ACS

# Quantized Exciton Motion and Fine Energy-Level Structure of a Single Perovskite Nanowire

Ying Tang, Xiaoyong Wang, et al. APRIL 01, 2022 NANO LETTERS

READ 🗹

READ 🗹

### Tip-Induced Strain Engineering of a Single Metal Halide Perovskite Quantum Dot

Hyeongwoo Lee, Kyoung-Duck Park, et al. MAY 14, 2021 ACS NANO

Light Propagation and Radiative Exciton Transport in Two-Dimensional Layered Perovskite Microwires

Yangzi Zheng, Anton V. Malko, *et al.* DECEMBER 29, 2020 ACS PHOTONICS

READ 🗹

### Effect of Anisotropic Confinement on Electronic Structure and Dynamics of Band Edge Excitons in Inorganic Perovskite Nanowires

 Brendan D. Folie, Naomi S. Ginsberg, et al.

 FEBRUARY 25, 2020

 THE JOURNAL OF PHYSICAL CHEMISTRY A

 READ Z

Get More Suggestions >